



# An "order N" method for the atomistic simulation of nano-oxides: Metal-supported MgO(100) islands

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#### **Oxide nano-objects: structure and properties**

Structural complexity driven by the environment & growth conditions



A large number of complex structures cannot be treated directly by ab initio simulations.

# **Metal-supported oxide 1ML**

#### FeO(111)/Pt(111)









Moiré structure due to the lattice mismatch (periodicity ~ 25 Å).

Strong rumpling of the oxide layer due to interface charge transfer. Modulation of the surface potential driven by the structure of the metal/oxide interface. Spontaneous charging and self-organization of adsorbates (Au monomers). Registry-dependent oxidation of oxide film.

Low temperature catalytic activity!

M. Ritter et al., PRB 57, 7240 (1998). JG, C. Noguera, PRB 79, 155433 (2009). E. Rienks et al., PRB **71**, 241404 (2005). L. Giordano et al., PRB **76**, 075416 (2007).

Nilius et al., PRL **95**, 066101 (2005). L. Giordano et al., PRL **101**, 026102 (2008).

Y.-N. Sun et al.,
J. Catal. 266, 359 (2009).
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JPC C 114, 21504 (2010).

#### MgO islands and films on transition/noble metal substrates

#### MgO(100)/Ag(100)



Schintke et al., 2004

#### MgO(100)/Mo(100)



Benia et al., 2010

#### MgO(100)/Ag(100)



Bieletzki et al., 2010

Cabailh et al., 2011

- interfacial strain due to lattice mismatch?
- formation of Moiré pattern & interface dislocation?
- magic sizes and shapes?

#### **Coupling between electronic and structural degrees of freedom**



Self-consistent treatment of the electronic structure may be necessary

### "Order N" method: background and performances

Semi-empirical Hartree-Fock quantum approach (INDO)





Eg., for 100 000 atoms: \* 5-10 min/iteration \* 8 GB RAM

100 000 years/iter. 7 TB RAM

# **Adjustment of parameters**

Case of MgO: adjustment to experimental and DFT results



# Transferability of the parameterization

Test of accuracy for unsupported MgO(001): films of increasing thickness



## Metal substrate: Potential energy surface

Case of weak metal-oxide interaction (eg., MgO/Ag(100)):

$$E^{sub} = \sum_{i=1}^{N} E_{int}^{at_i}(x_i, y_i, z_i)$$

$$E_{int}^{at}(x, y, z) = a(z) + b(z) \left[\cos(\frac{2\pi x}{a}) + \cos(\frac{2\pi y}{a})\right] + c(z) \left[\cos(\frac{2\pi (x+y)}{a}) + \cos(\frac{2\pi (x-y)}{a})\right]$$



Effect of the substrate on the island's structure



- Moiré pattern due to lattice mismatch;
- local lattice distortion (Mg-O distances): coincidence zones: bond expansion interfacial dislocations: bond contraction

Displacement of ions along x axis:

 $x_n = n\gamma a + \alpha a \sin 2\pi n (1 - \gamma) + \delta x_n$ 

 $\gamma a =$  mean lattice parameter  $\gamma = 7/8$ : 8 oxygen rows for 7 metal rows  $\delta x_n =$  edge effects

Dislocation network lattice parameter:

 $A = \frac{\gamma a}{1 - \gamma}$ 

In the present case (a=3.2 A, F=2):  $\gamma a \sim a_{MgO unsupported}$ 

Periodicity determined by the lattice misfit at the interface.

Effect of the substrate on the island's electronic characteristics





Coincidence zones: expanded bonds, lower electrostatic potential, smaller gap

STM observation of Moiré pattern MgO/Mo(001)



Fig. 5: (a): STM bias series of an identical MgO/Mo(001) region  $(100 \times 100 \text{ nm}^2)$ . The dashed lines mark selected domain boundaries and should serves as guides to the eye. The characteristic square pattern is only visible in a small bias range around 4.0 V. (b) dz/dV spectra taken above a bright and a dark region of the coincidence pattern. The first FER is clearly down-shifted above the bright lines compared to the center of the coincidence cell.

H.M. Benia, P. Myrach, N. Nilius, H.-J. Freund , Surf. Sci. 604, 435 (2010)

Effect of the substrate induced constrain on the preferential position of point defects





Benia et al. 2010

adsorbed  $(MgO)_2$ 

# Exploring phase diagram (misfit, interaction strength): 1D chains

Numerical solution for a linear  $(MgO)_{60}$  chain



- di-atomic periodic unit,
- vertical relaxation of all atoms,
- "realistic" border effects.

# Exploring phase diagram (misfit, interaction strength): 2D islands



Oxide/substrate misfit

Oxide-substrate interaction strength



BUT: Spontaneous tetragonal distortion at small size (N=900)



#### Stability as a function of size: magic islands

 $\Delta^2 E(N) = E(N+1) + E(N-1) - 2E(N) > 0$ 



$$L = \frac{1}{\sqrt{2}} \frac{\gamma a}{1 - \gamma} = \frac{A}{\sqrt{2}}$$

#### MgO/Ag(100)



Cabailh et al. 2011 Average island size: 80±10Å Schintke et al. 2004 Average island size 70Å



Ag: a=2.832 Å MgO: a<sub>I</sub>=2.824 Å Consistent with  $\gamma a \sim a_I$ 



#### MgO/Mo(100)

Benia et al. 2010 Average island size: 45-55 Å

Mo: a=3.147 Å MgO:  $a_I$ =2.824 Å Consistent with  $\gamma a > a_I$ 

#### Substrate-induced patterning of island edges







#### Growth shapes: MgO/Ag(100)





Cabailh et al. 2011

# Summary

- \* Semi-empirical atomistic "order N" approach for large scale calculations.
- \* Application to (medium size) MgO(100) islands on a metal substrate:
  - Potential Energy Surface method
  - structure of the islands (Moiré, interfacial dislocations,

commensurability locking, edge effects)

- electronic structure (inhomogenity inside the islands, edge effects)
- size effects:
  - smallest sizes, mainly driven by edge effects
  - larger sizes: oscillatory behavior related to introduction of dislocations
  - magic islands
- phase diagram (interaction, misfit)
- \* Perspectives
  - beyond PES approximation
  - equilibrium/growth shapes
  - adsorption properties

Shape transition Anatase TiO<sub>2</sub>(001)/SrTiO<sub>3</sub>(001) Marshall and Castell 2009

