

Plasticité des minéraux du manteau terrestre

Philippe Carrez

Unité Matériaux et Transformations

CNRS UMR 8207

Université de Lille 1

Villeneuve d'Ascq

Le laboratoire



se décline en quatre équipes :

Matériaux Moléculaires et
Thérapeutiques

Métallurgie Physique et Génie des
Matériaux

Physique des Minéraux

Ingénierie des Systèmes Polymères.

Equipe : Métallurgie Physique

Modélisation multi-échelle de matériaux métalliques au voisinage ou loin de l'équilibre thermodynamique, en présence ou non de sollicitations extérieures (chocs, irradiation, contraintes mécaniques)

Permanents :

A. Legris

R. Besson

L. Thuinet

C. Becquart

Post doctorants :

A. De Backer

J. Boisse

J.B. Piochaud

Doctorants :

H. Rouchette

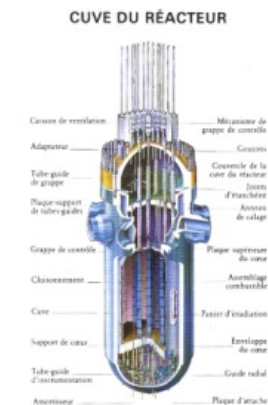
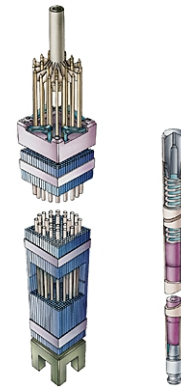
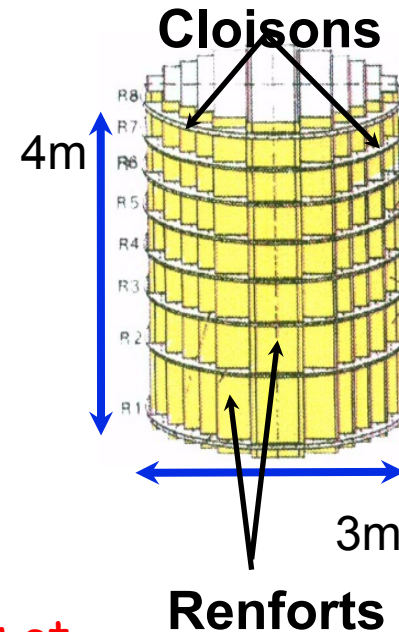
J. Kwon

-Alliages de Zr

-Aciers ferritiques et austénitiques

-W

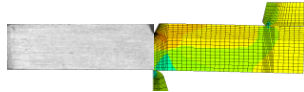
-NiAl



Equipe : Métallurgie Physique

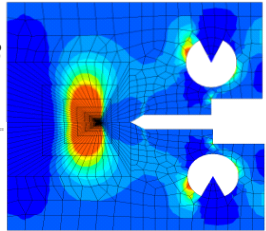
m^3

10 years

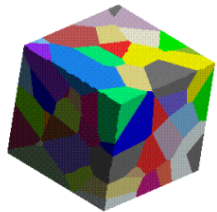


Finite elements

cm^3



Micro-macro



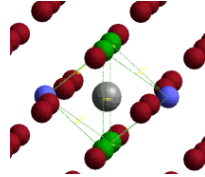
μm



Dislocation dynamics

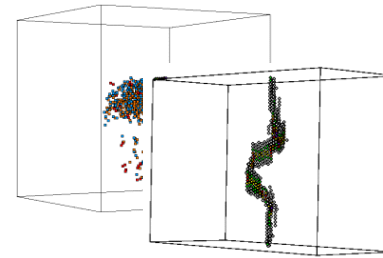
ab initio, dft

$1nm^3$, 0-ps

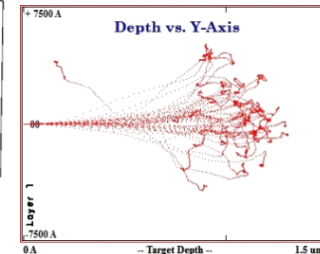


Molecular dynamics

$(10 nm)^3$, ns



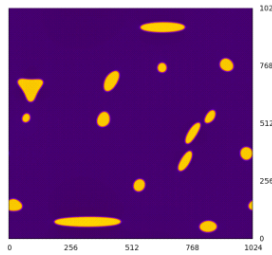
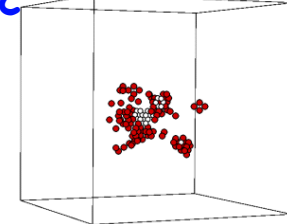
Binary collision approximation



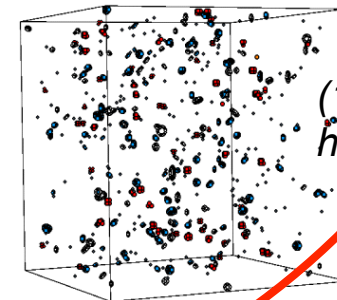
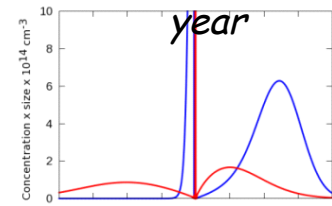
- Alliages de Zr
- Aciers ferritiques et austénitiques
- W
- NiAl

Akmc

(nm) , s-h

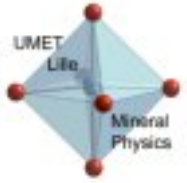


Phase Field Rate theory



$(100 nm)^3$
hour-year

Okmc



Equipe : Physique des Minéraux

=> astrominéralogie

=> **plasticité** des minéraux sous haute pression

Permanents :

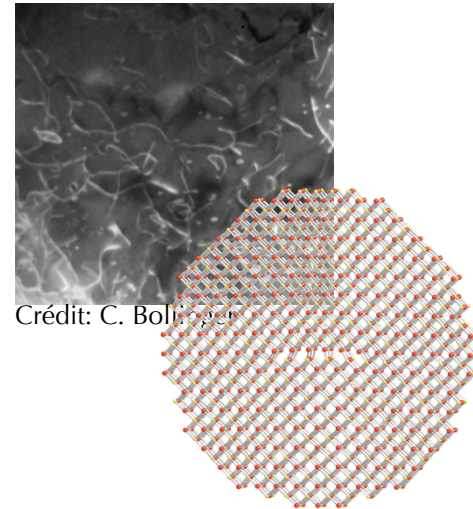
Ph. Carrez, P. Cordier, K. Gouriet

Post doctorants :

P. Hirel, F. Boioli

Doctorants :

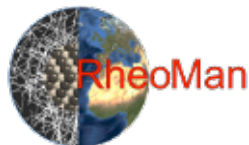
A. Kraych, S. Ritterbex

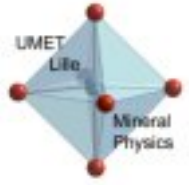


Crédit: C. Bollinger

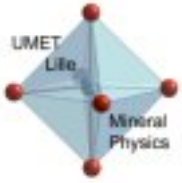


Projet ERC Adv. Grant

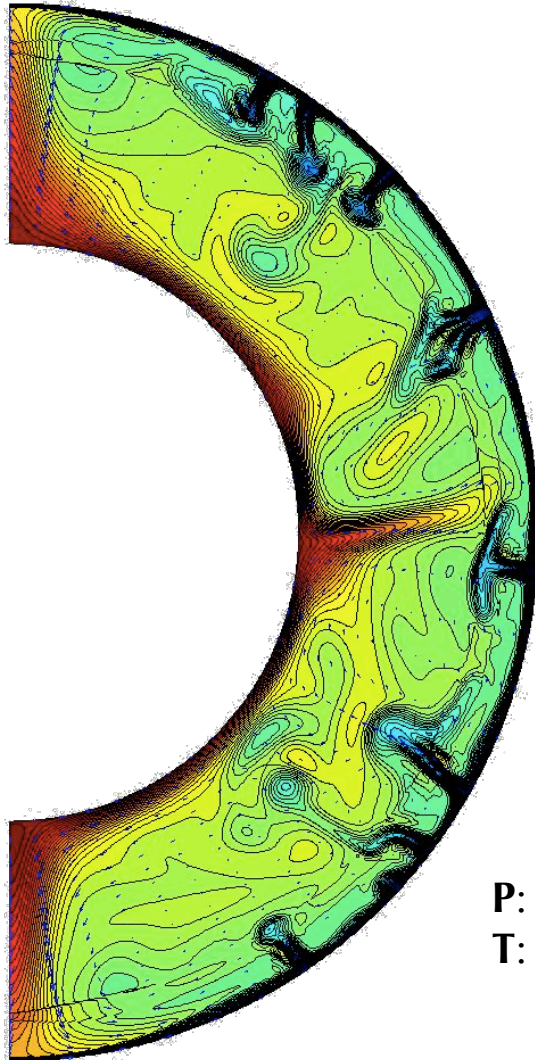




Plasticité des minéraux du manteau terrestre

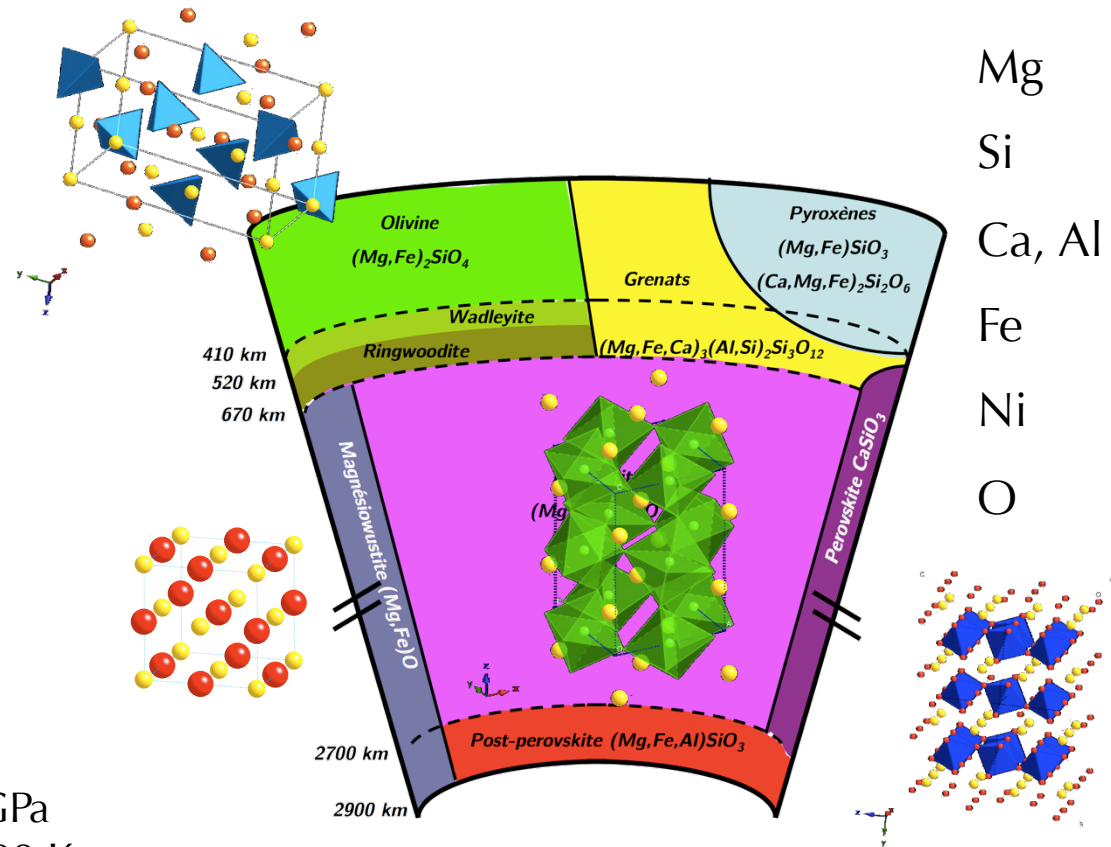


Convection et deformation

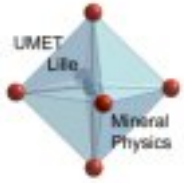


Crédit: S. Bulter

P: 10 – 130 GPa
T: 1000 – 3000 K

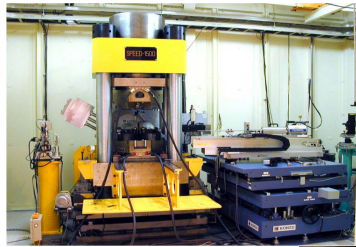


Mécanismes de déformation ?

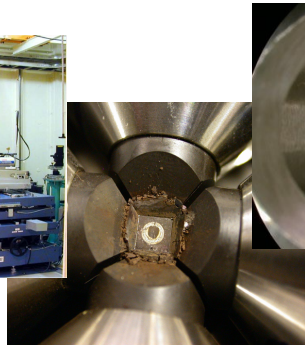


La plasticité des minéraux: du point de vue expérimental

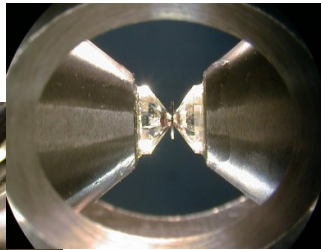
Conditions (P,T,ε̇) sont particulièrement difficiles à reproduire dans les conditions du laboratoire



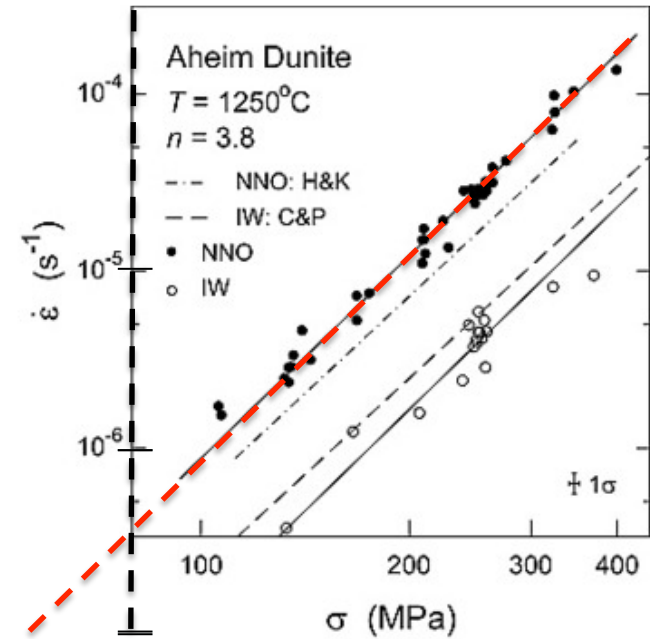
Presse Gros Volume
P_{max} = 20-80 GPa



D-Dia
P_{max} = 20 GPa



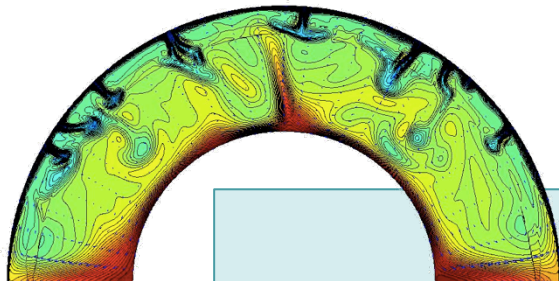
Diamant anvil cell (DAC)
P_{max} = 400 GPa

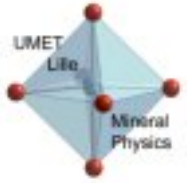


Mackwell et al. 2008

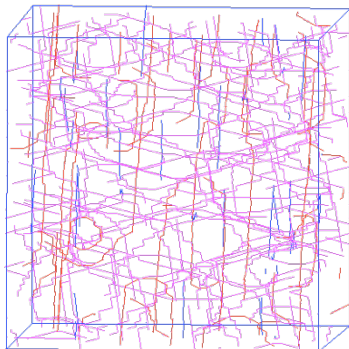
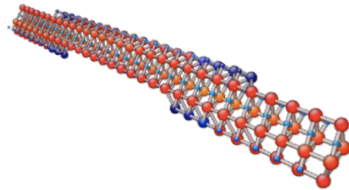
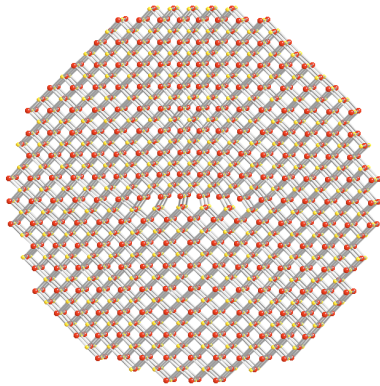
$$\dot{\epsilon} = \dot{\epsilon}_0 \cdot \sigma^n \cdot \exp[-Q/RT]$$

$$Q = E + PV$$

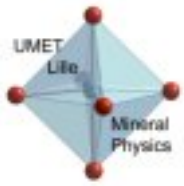




L'approche numérique: Objectifs

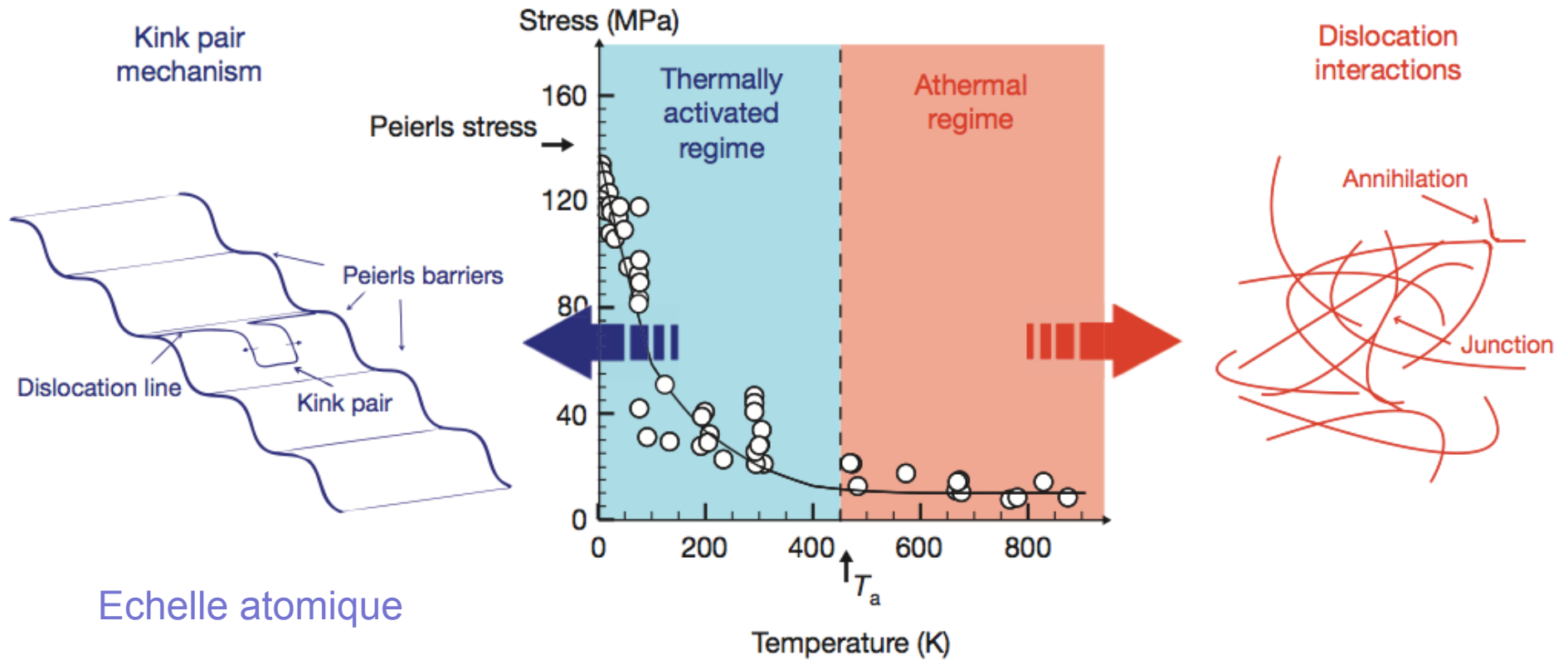


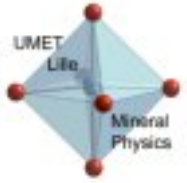
- . Structures de cœur de dislocations dans les minéraux du manteau
- . Friction de réseau et Contrainte de Peierls
- . Effet de pression
- . Effet de température et mobilité
- . Fluage



Modélisation des CRSS

Echelle mésoscopique / code DD





DFT/calculs par potentiels

DFT



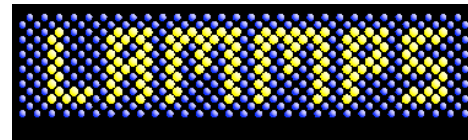
Taille des volumes simulés / nombre d'atomes dans les mailles élémentaires (> 20)

Calcul de γ -surface (énergie de faute d'empilement généralisé)

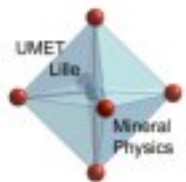
Approche semi-continu de type Peierls-Nabarro

Potentiels

$$V_{ij}(r_{ij}) = \frac{q_i q_j}{r_{ij}} + A_{ij} \exp(-r_{ij}/\rho_{ij}) - \frac{C_{ij}}{r_{ij}^6}$$

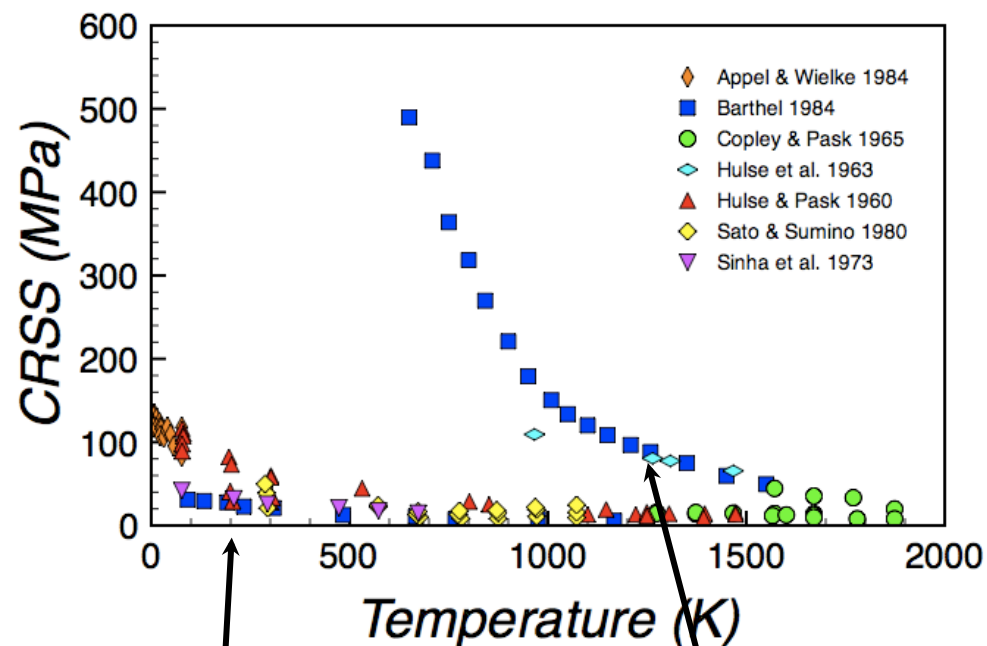
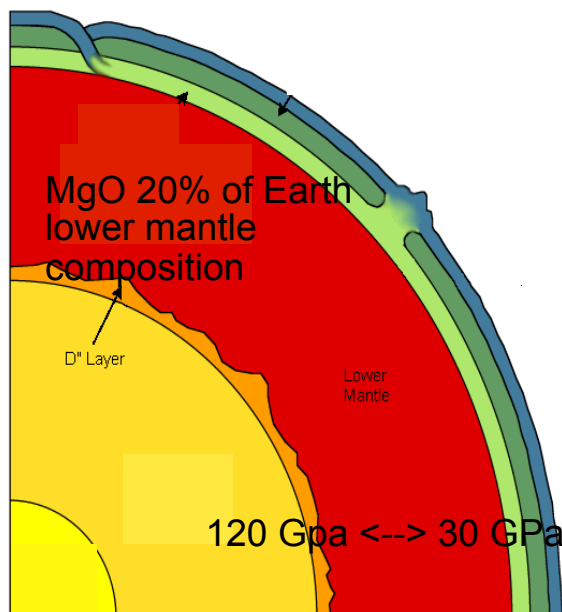
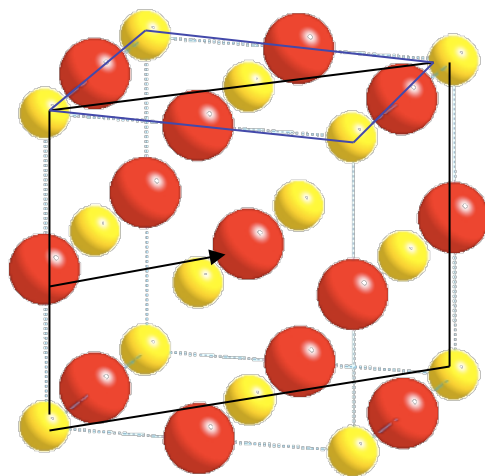


GULP



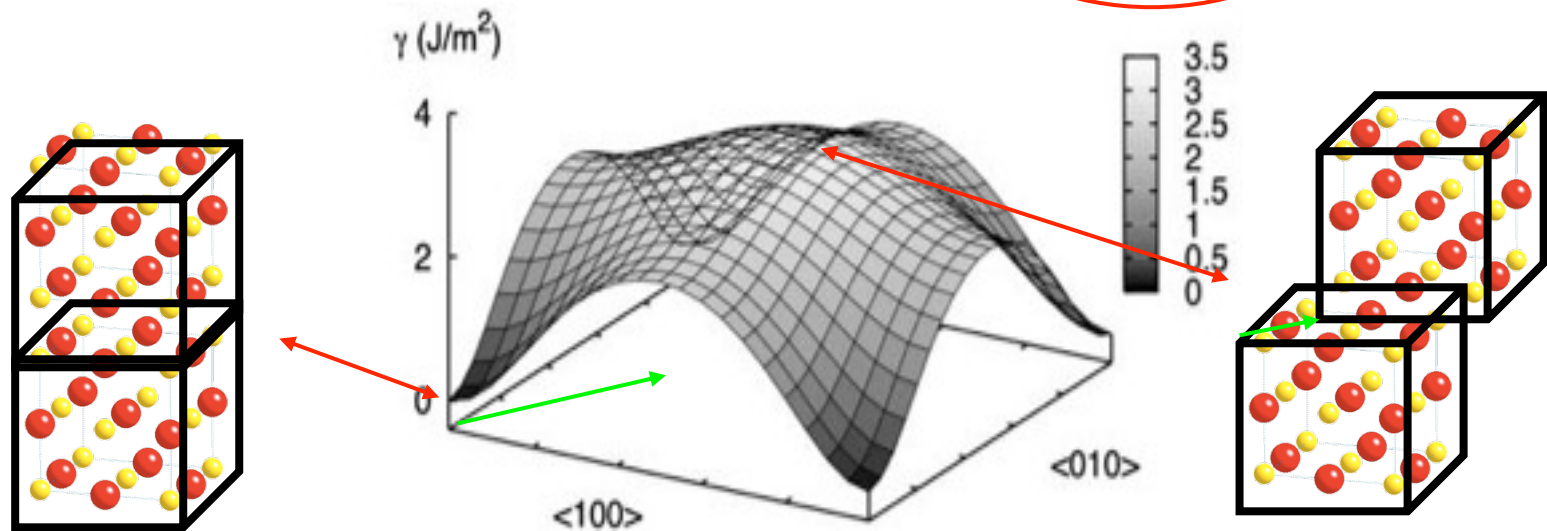
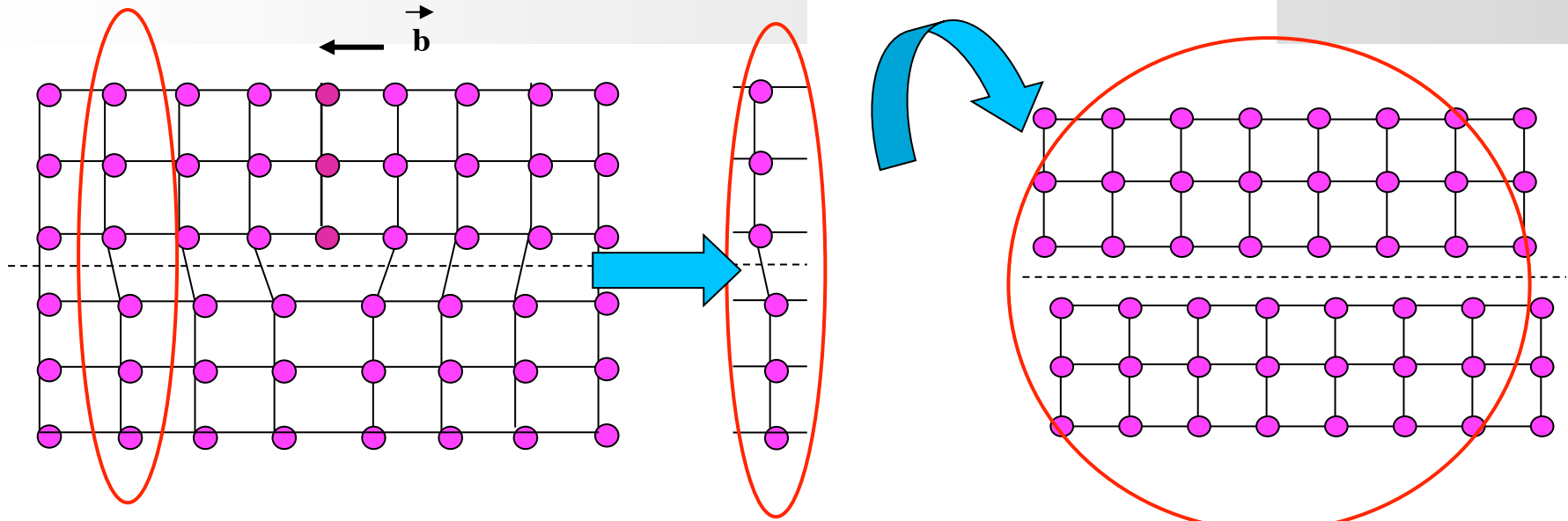
Illustration

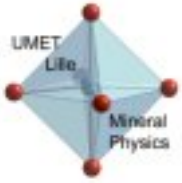
dislocation vis de vecteur de Burgers $\frac{1}{2}\langle 110 \rangle$ dans MgO



$\frac{1}{2}\langle 110 \rangle \{110\}$ and $\frac{1}{2}\langle 110 \rangle \{100\}$

Faute d'empilement généralisée et γ -surface

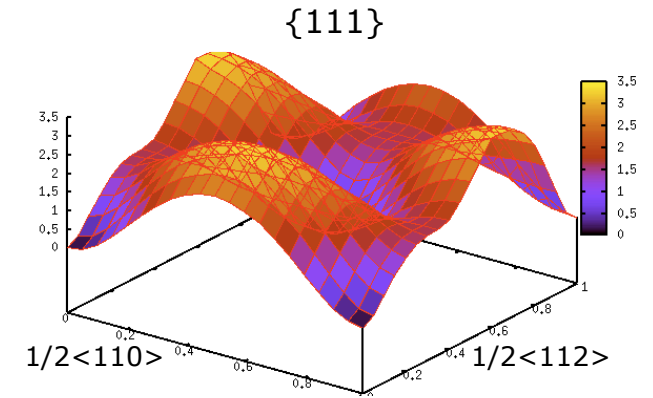
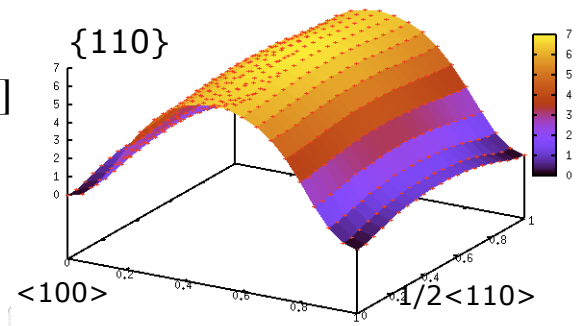
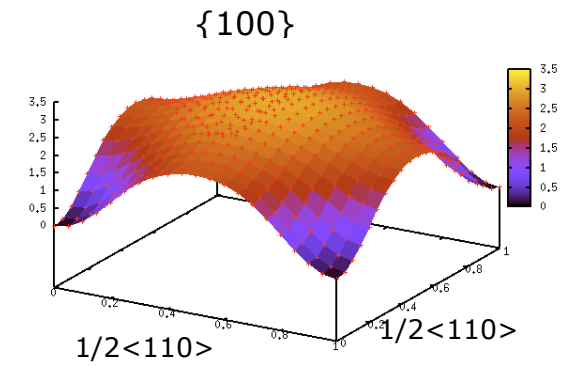
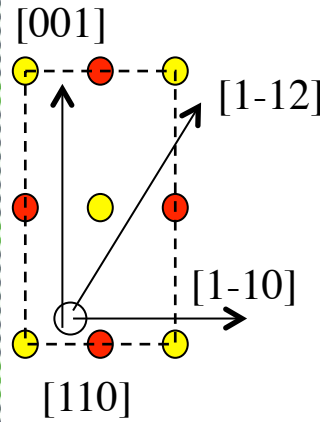
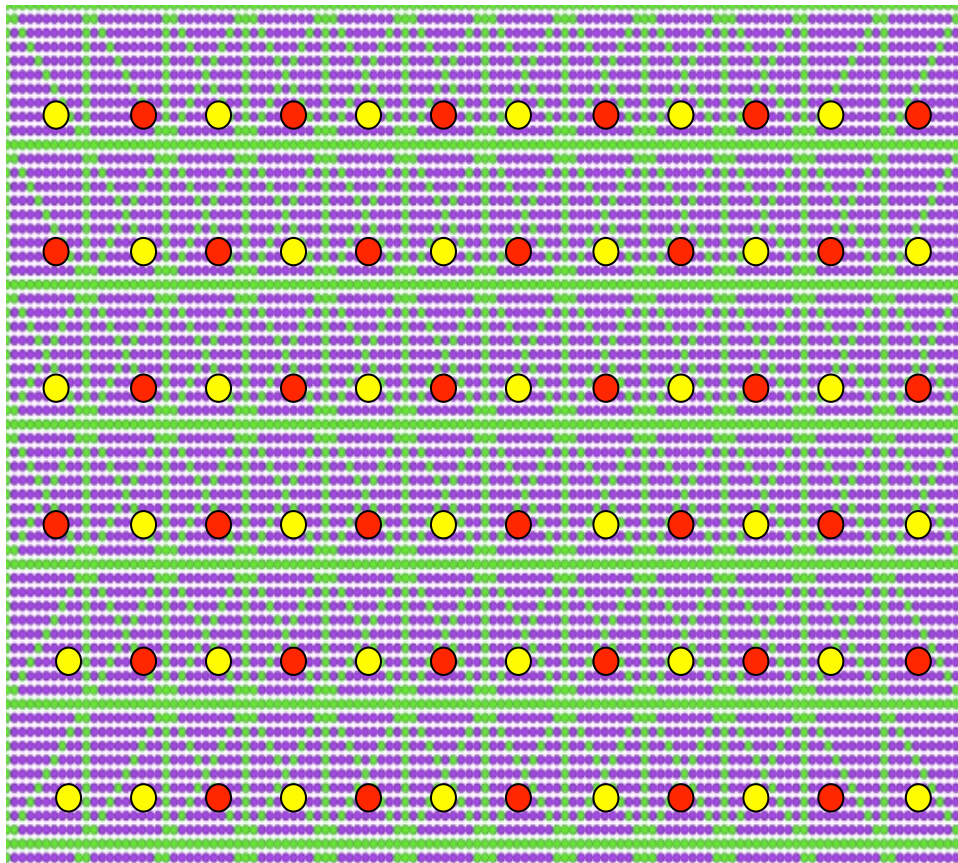




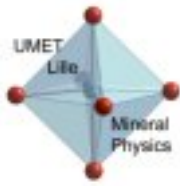
Le modèle de Peierls-Nabarro-Galerkin

dislocation vis de vecteur de Burgers $\frac{1}{2}\langle 110 \rangle$

$$\int_V \left\{ E^e [u, S] + \frac{1}{2} \Omega \dot{u}^2 \right\} dV + \int_{\Sigma} E^{isf} [S] d\Sigma$$

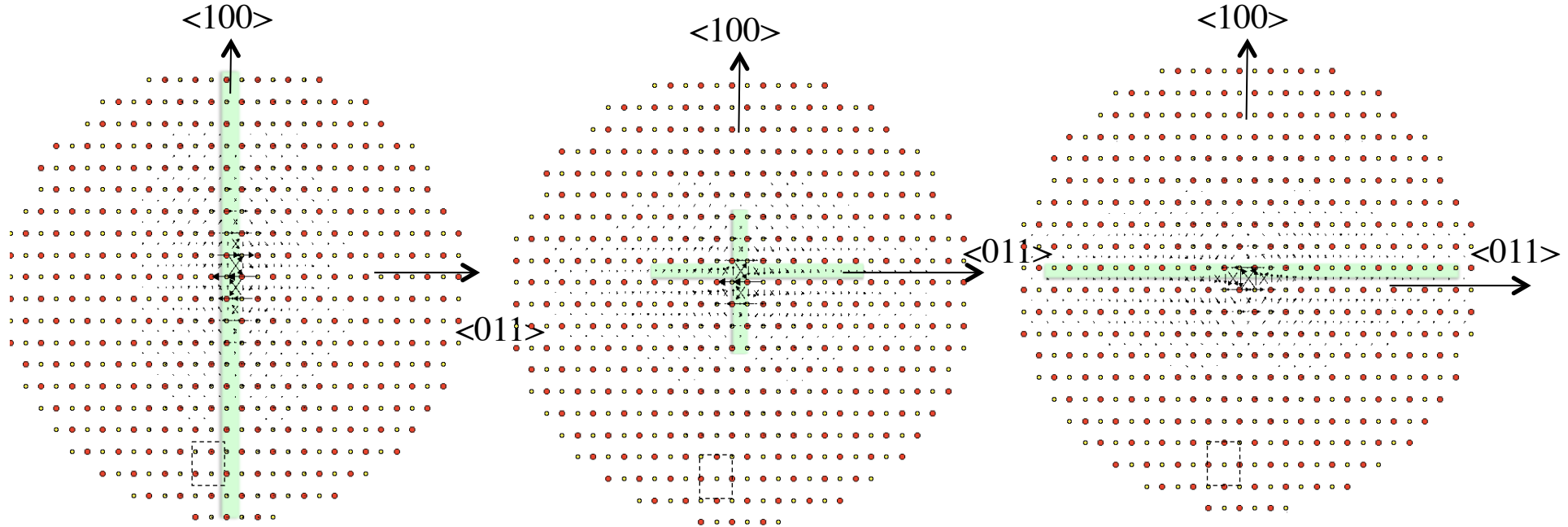


Code de calcul Cod²ex développé au CEA/DAM



L'effet de la pression sur la structure de cœur de la dislocation vis $\frac{1}{2}\langle 110 \rangle$ dans MgO

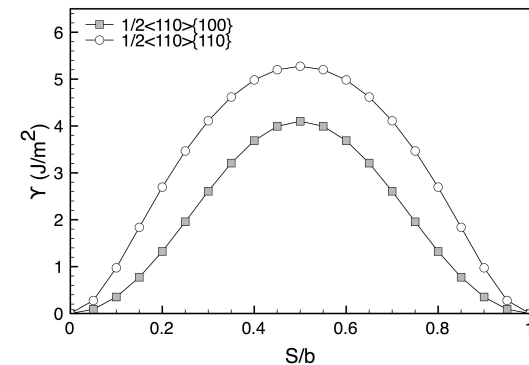
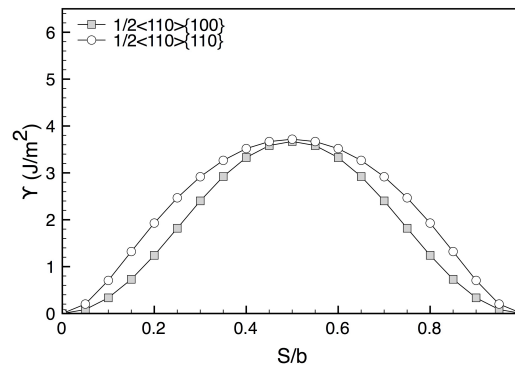
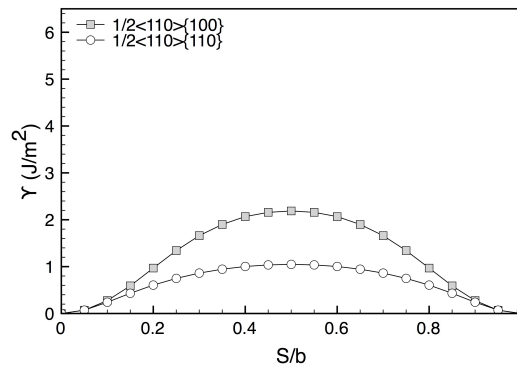
Evolution de la structure de cœur d'une dislocation vis de vecteur de Burgers $\frac{1}{2}\langle 110 \rangle$ dans MgO

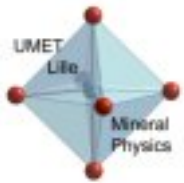


P=0 GPa

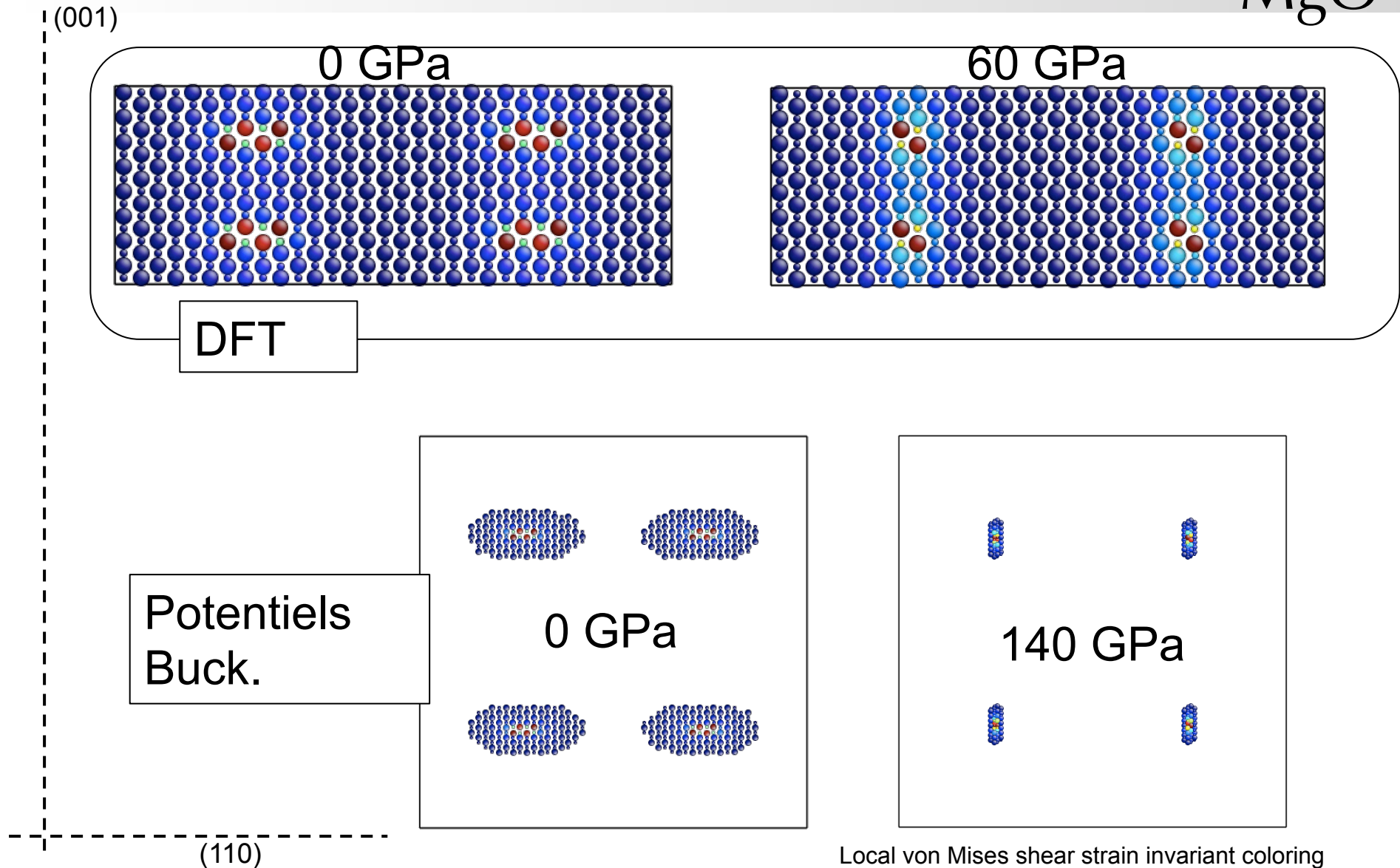
P=60 GPa

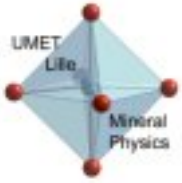
P=100 GPa



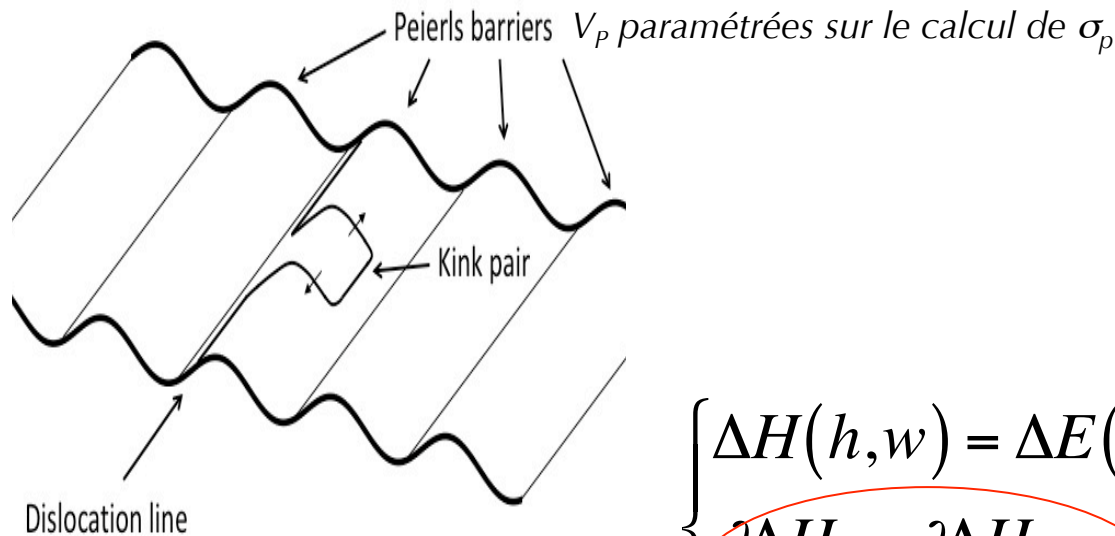


L'effet de la pression sur la structure de coeur de la dislocation vis $\frac{1}{2}\langle 110 \rangle$ dans MgO



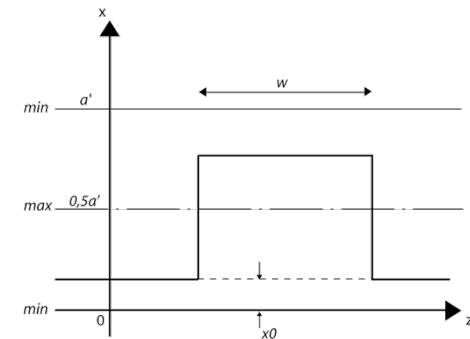


Enthalpie critique de nucléation d'un double décrochement

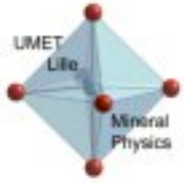


$$\begin{cases} \Delta H(h, w) = \Delta E(h, w) + \Delta P(h, w) - W(h, w) \\ \frac{\partial \Delta H}{\partial h} = \frac{\partial \Delta H}{\partial w} = 0 \end{cases}$$

- Elastic Interactions Theory (Koizumi et al., 1993)

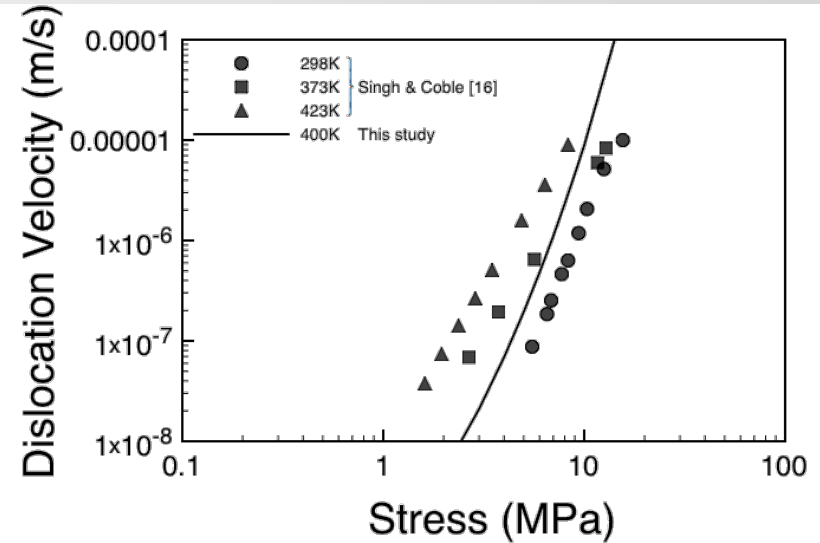
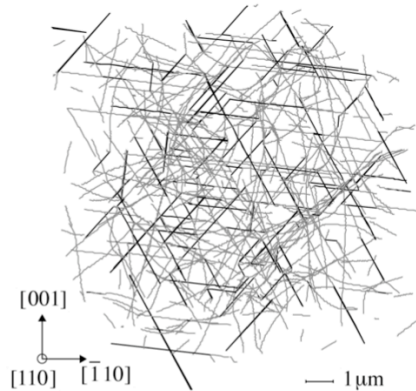


$$h^*, w^*, \Delta H^*$$



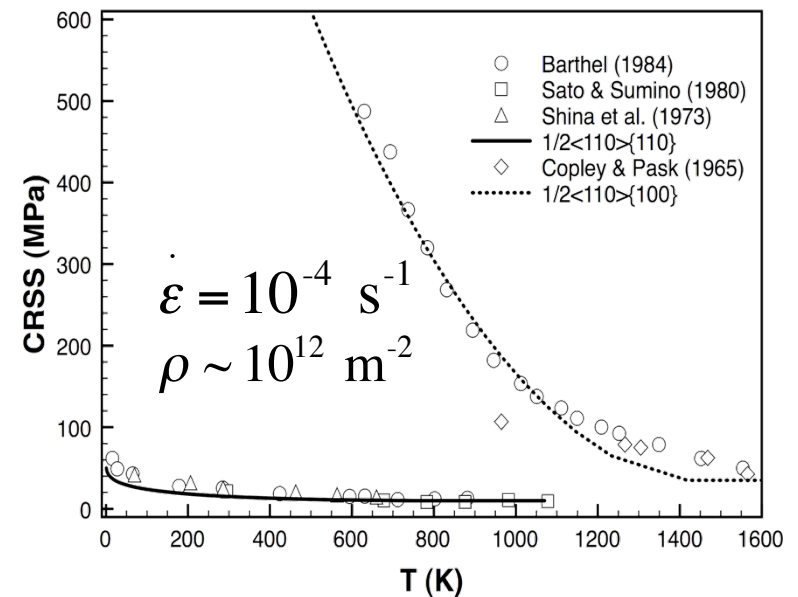
Confrontation aux données expérimentales

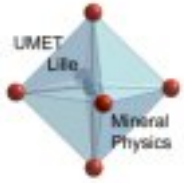
$$v = a' \cdot \frac{L}{w^*(\tau)} \cdot \frac{v_D b}{w^*(\tau)} \cdot \exp\left(-\frac{\Delta H^*(\tau)}{kT}\right)$$



Loi d'Orowan $\dot{\epsilon} = \rho b v$

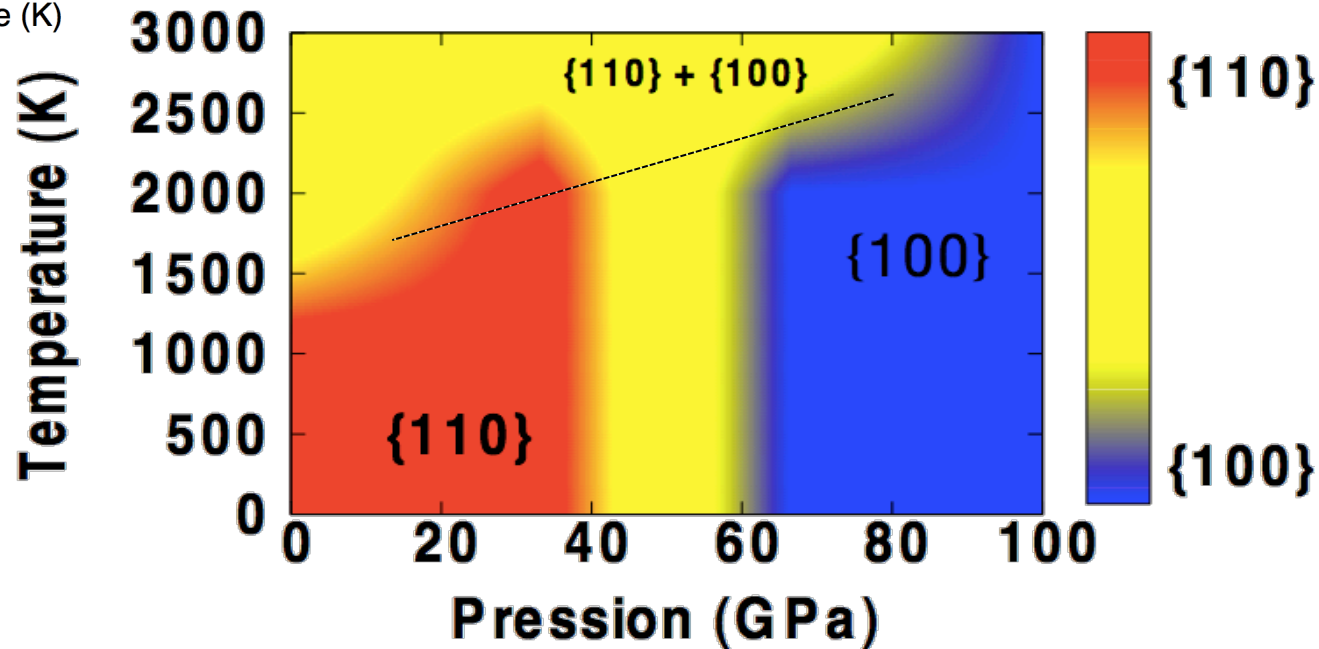
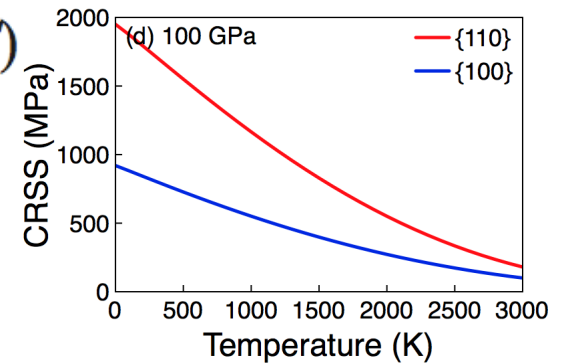
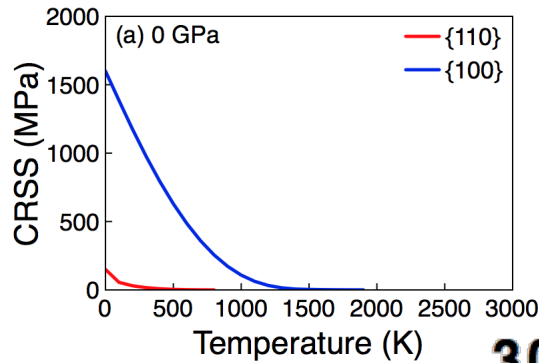
$$\dot{\epsilon} = \sqrt{\rho} \frac{v_D a' b^2}{2w_c^2} \exp(-\Delta H^*(\tau)/kT)$$



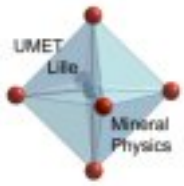


Evolution des CRSS de MgO en fonction de la pression

$$\dot{\epsilon} = \sqrt{\rho} \frac{v_D d' b^2}{2w_e^2} \exp(-\Delta H^*(\tau)/kT)$$

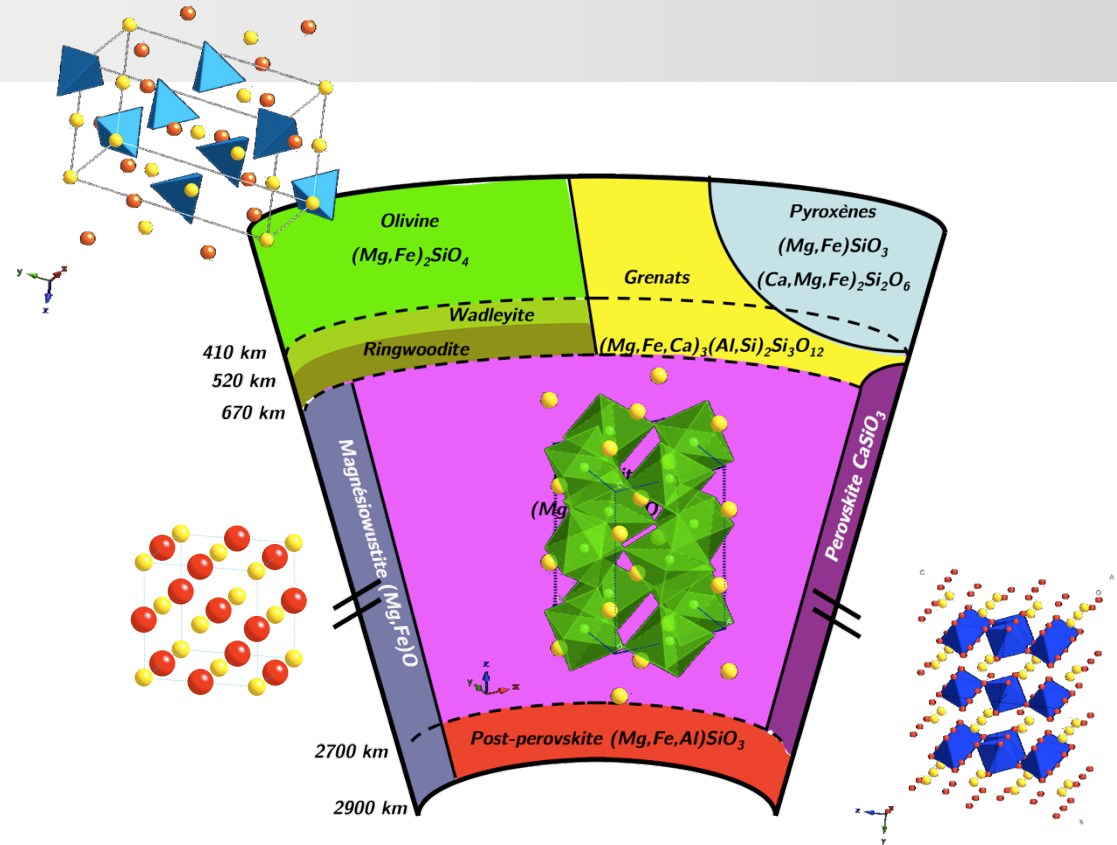
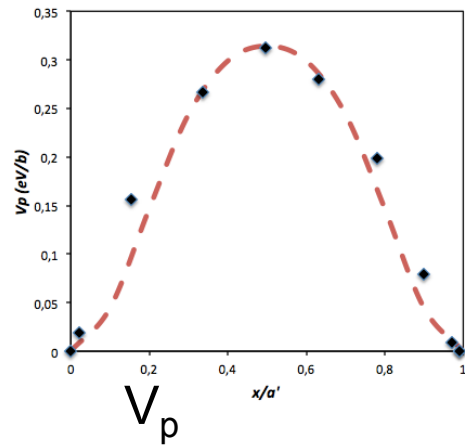


Confirmation expérimentale -> Girard et al. (2012)

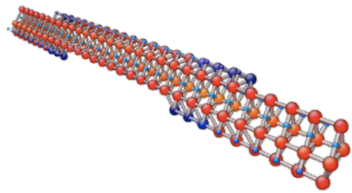


Conclusion

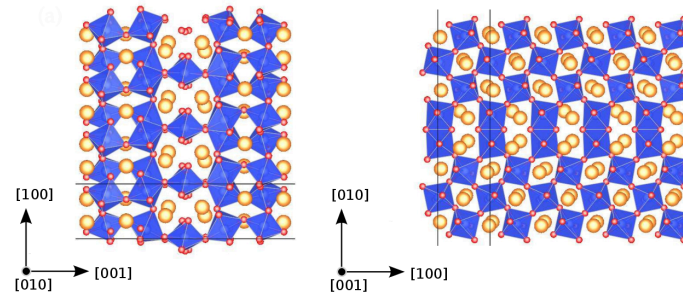
NEB

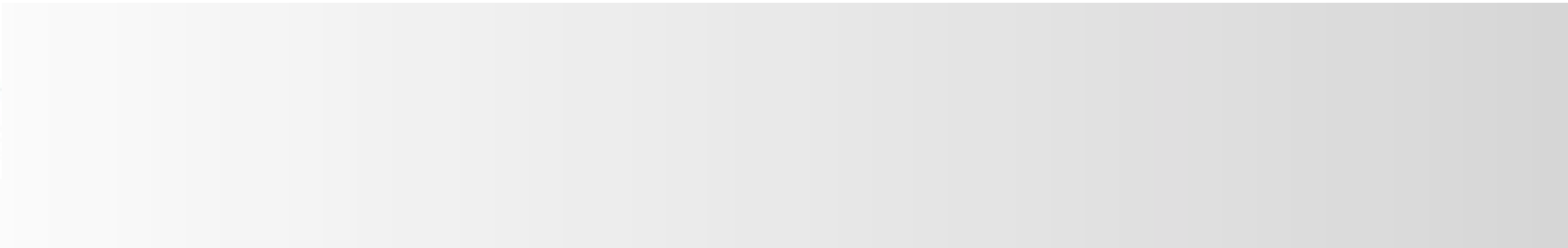
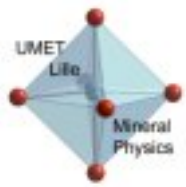


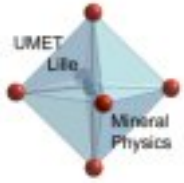
Perovskite: vis [100]



$\Delta H(\tau)$

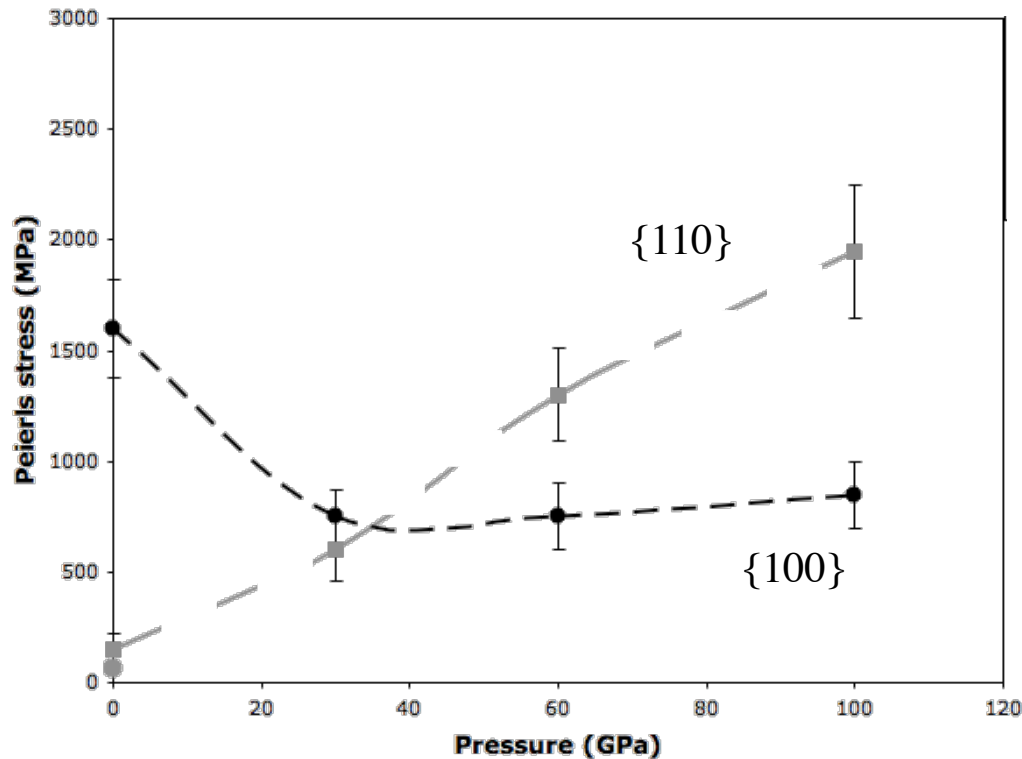






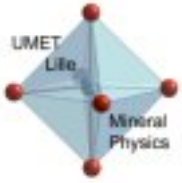
L'effet de la pression sur la structure de coeur de la dislocation vis $\frac{1}{2}\langle 110 \rangle$ dans MgO

A P=0 GPa, étalement du coeur de la dislocation dans $\{110\} \Rightarrow \sigma_p^{\{110\}} \ll \sigma_p^{\{100\}}$

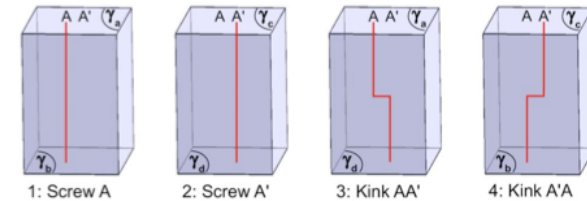
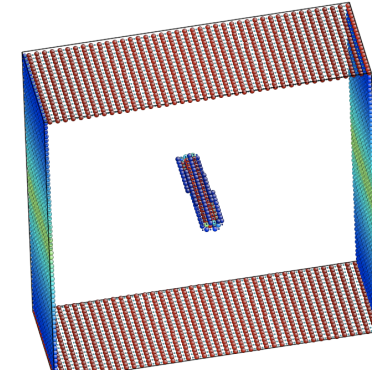
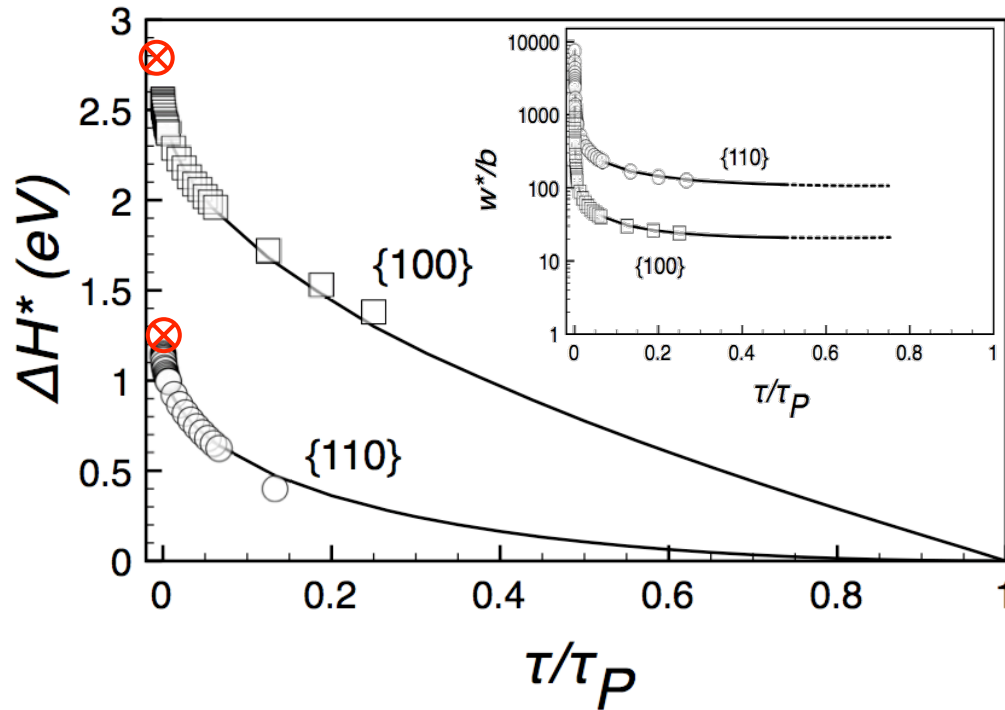


A P=100 GPa, étalement du coeur de la dislocation dans $\{100\}$

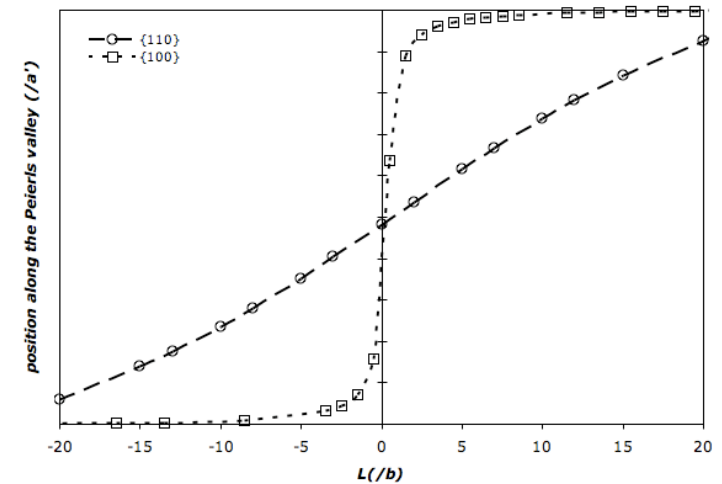
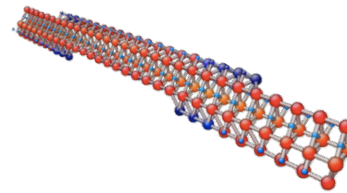
$$\Rightarrow \sigma_p^{\{100\}} < \sigma_p^{\{110\}}$$

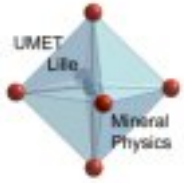


Enthalpie critique de nucléation d'un double décrochement



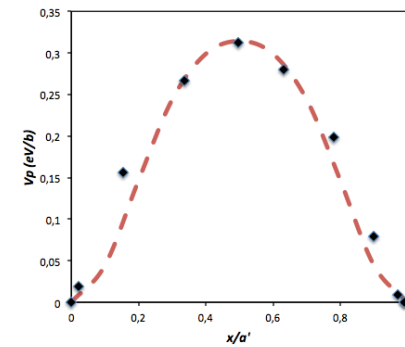
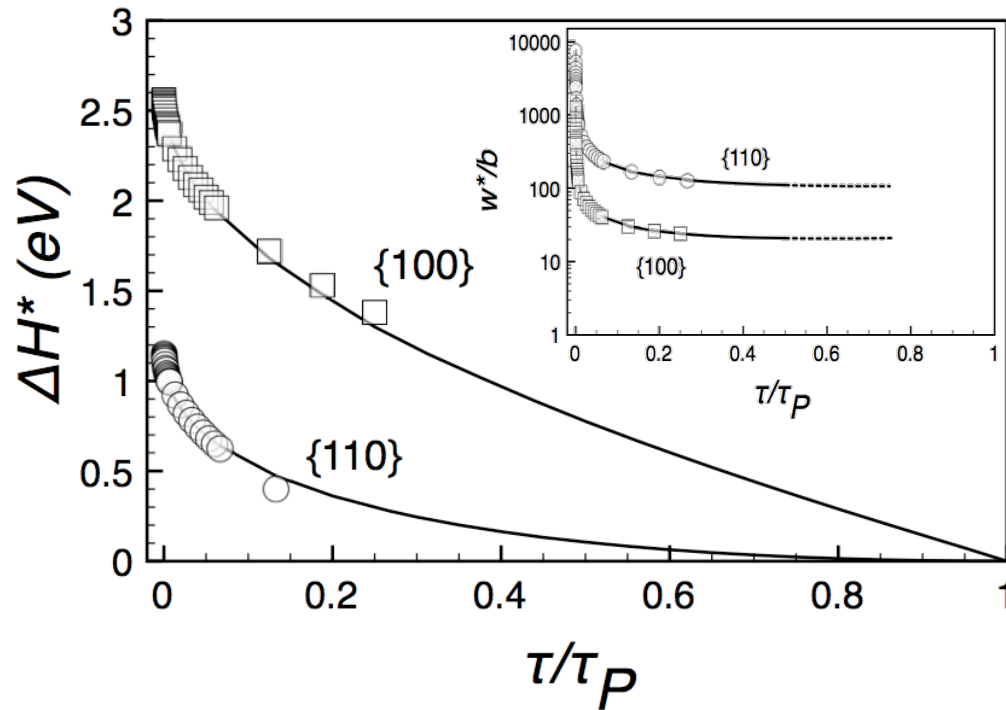
Pizzagalli et al. (2008)



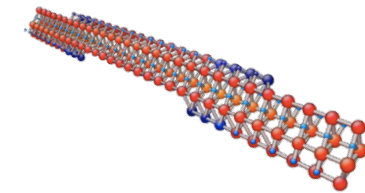


Enthalpie critique de nucléation d'un double décrochement

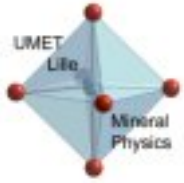
NEB+climbing image



V_p

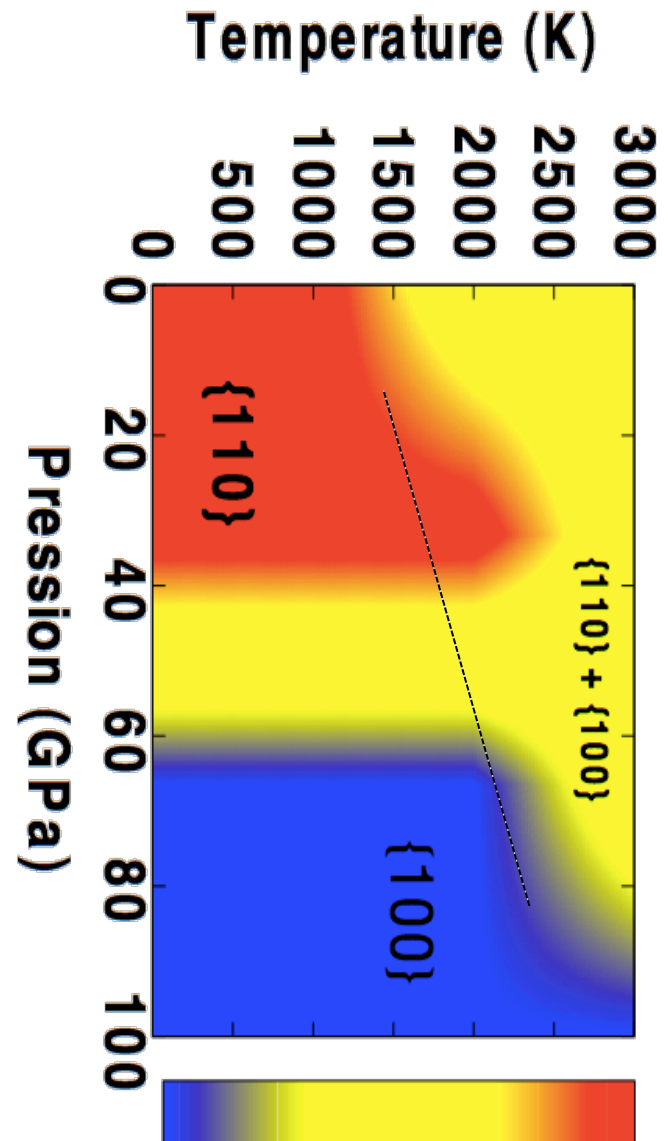
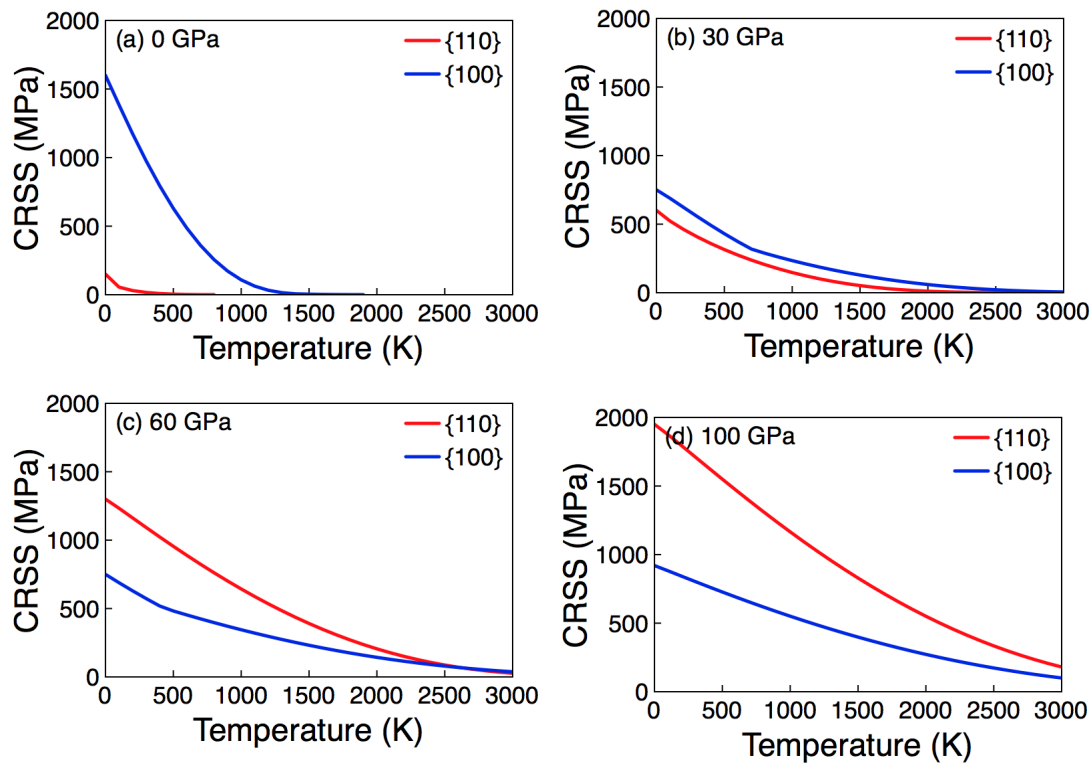


$\Delta H(\tau)$

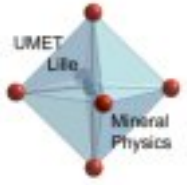


Evolution des CRSS de MgO en fonction de la pression

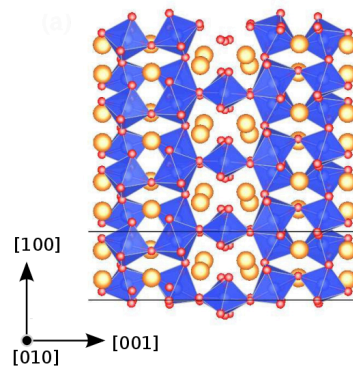
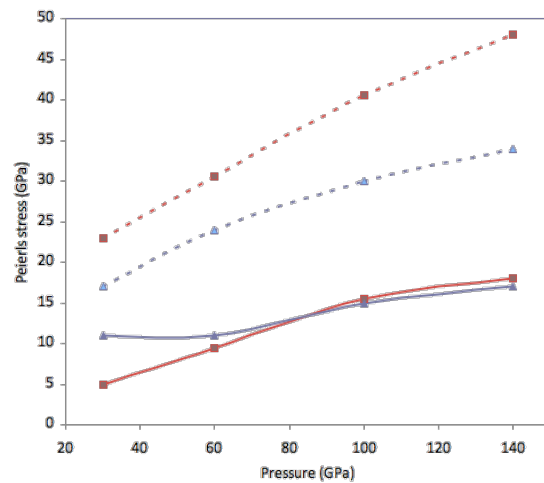
$$\dot{\epsilon} = \sqrt{\rho} \frac{v_D d' b^2}{2w_e^2} \exp(-\Delta H^*(\tau)/kT)$$



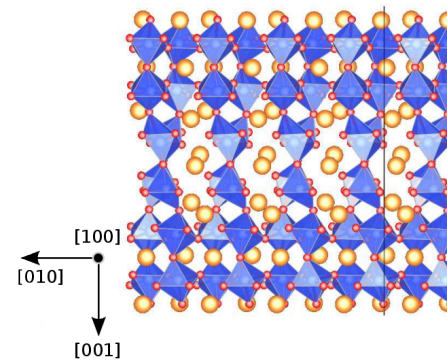
Amodeo et al. (2012)



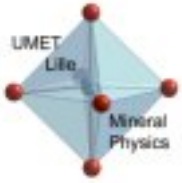
Partie III: Calculs par potentiels



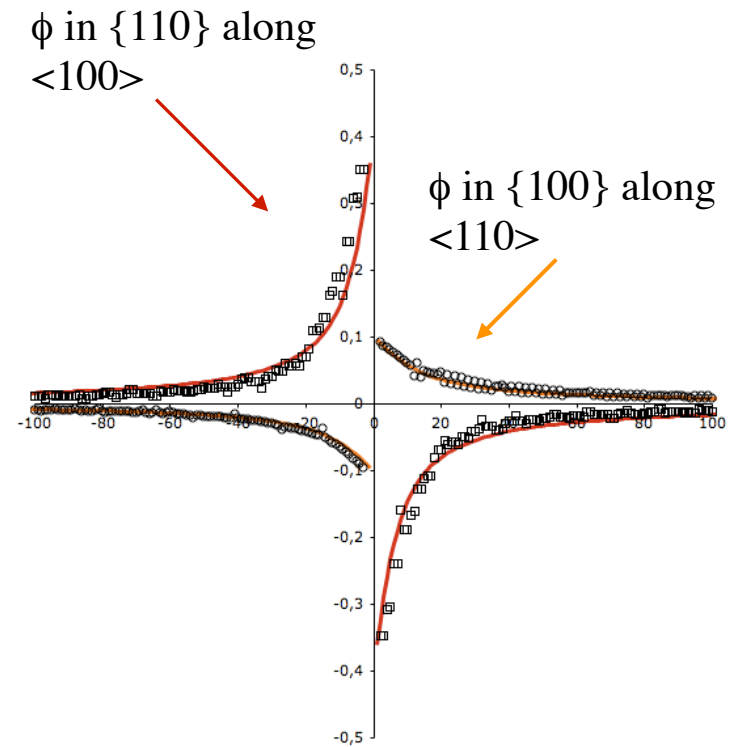
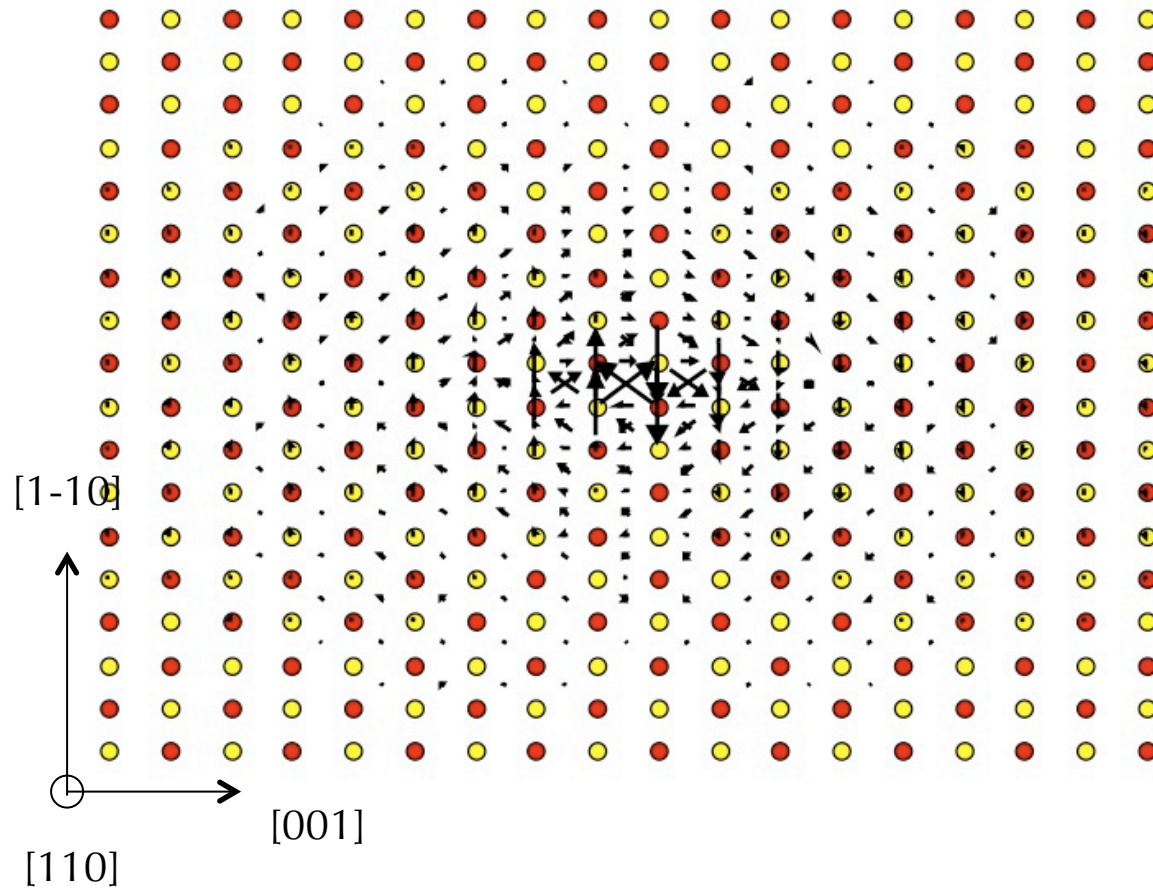
Perovskite MgSiO_3

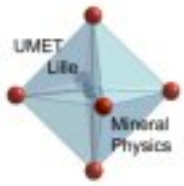


Thèse A. Kraych



$1/2\langle 110 \rangle$ screw core structure





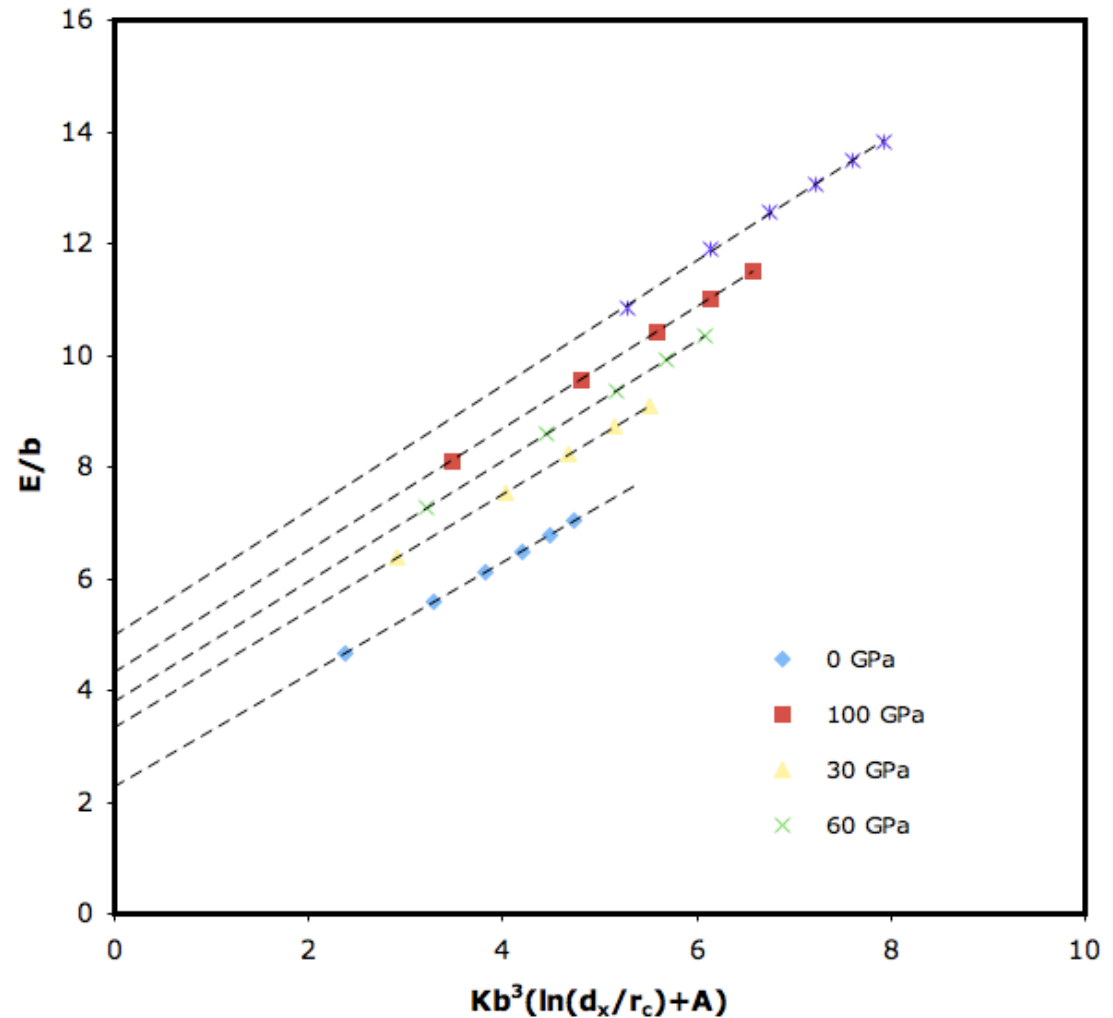
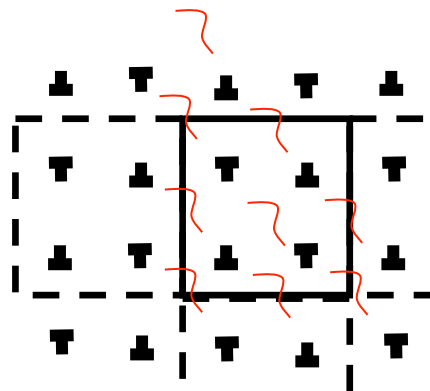
Quadrupole energy

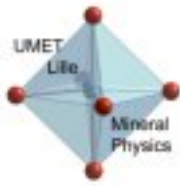
$$W_{1-2}/L = -\frac{\mu b_1 b_2}{2\pi} \ln(R/R_a) = \pm 2Kb^2 \ln(d/r_c)$$

In a quadrupole, energy per dislocation per b is then given by

$$E = E_c(r_c) + Kb^3(\ln(d/r_c) + A)$$

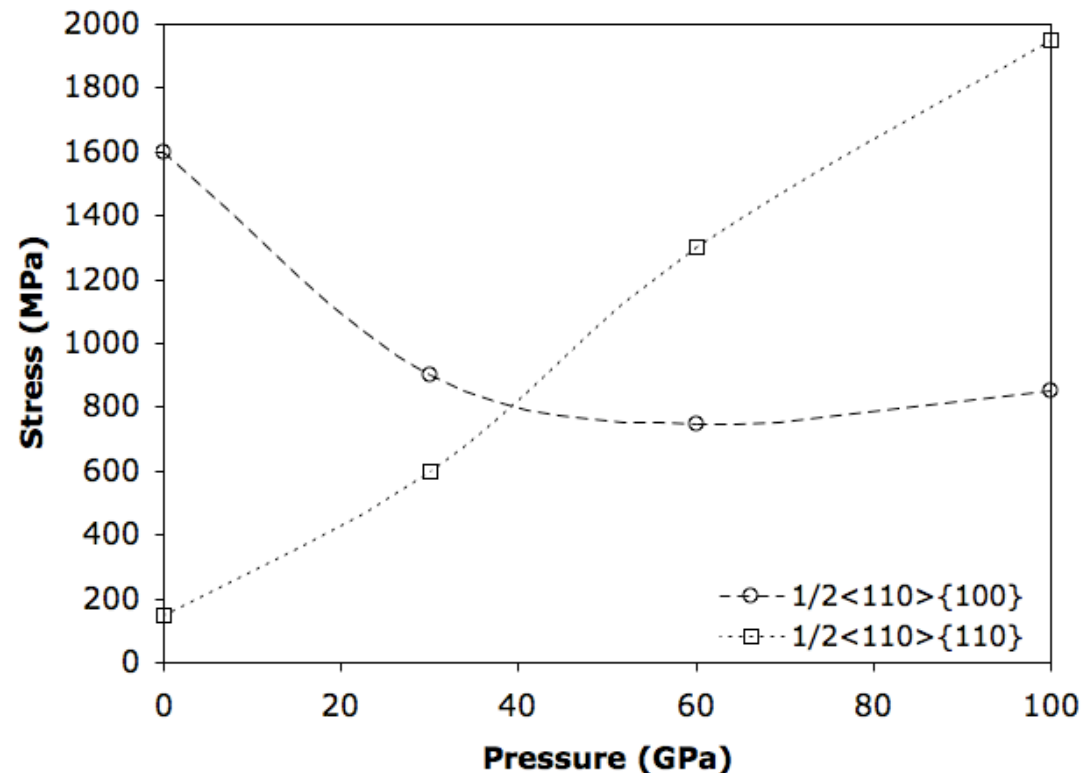
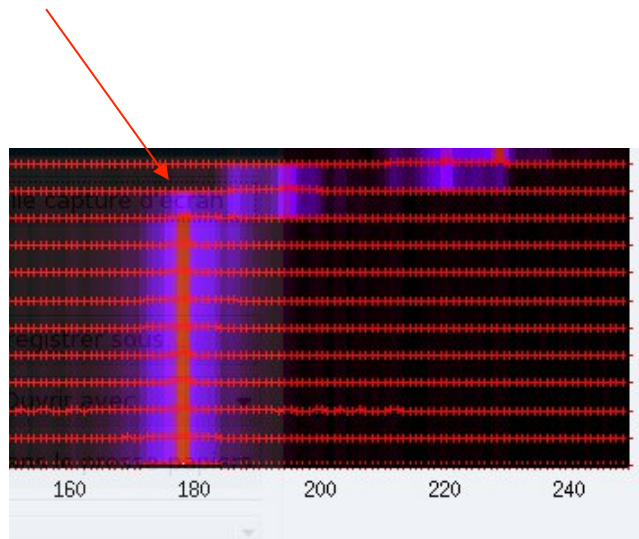
A contains all the effect of the infinite sums of dislocation interactions ($A=0.545$)

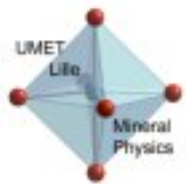




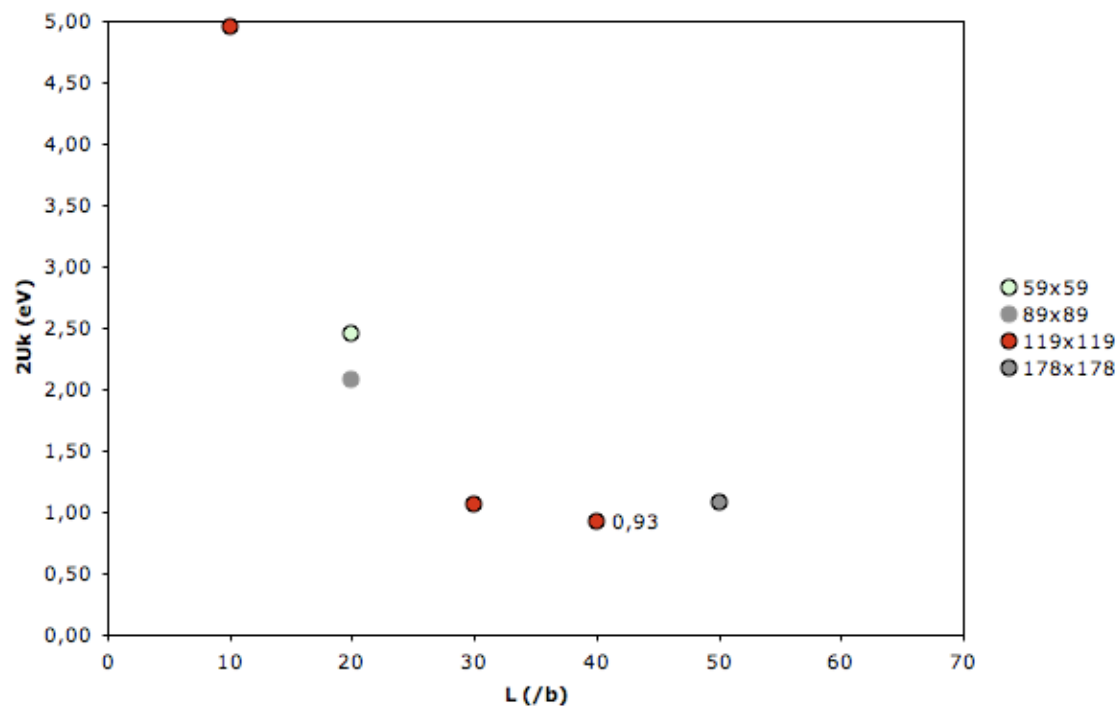
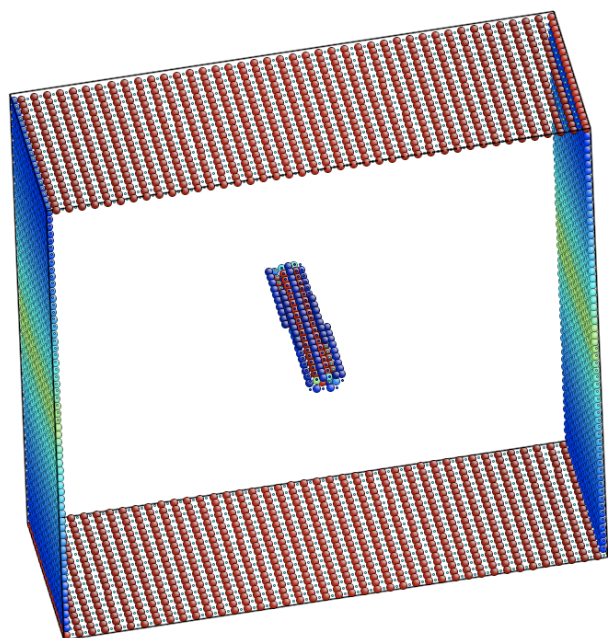
Effect of Pressure on Peierls stresses of screw dislocation

As core spreads from $\{110\}$ to $\{100\}$, Peierls stresses evolve and ones may expect a change in favourable slip system from $1/2\langle 110\rangle\{110\}$ to $1/2\langle 110\rangle\{100\}$

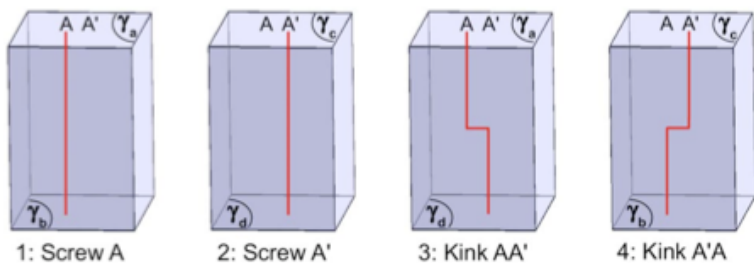




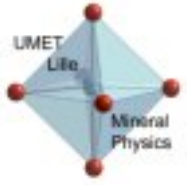
Isolated kink and U_k



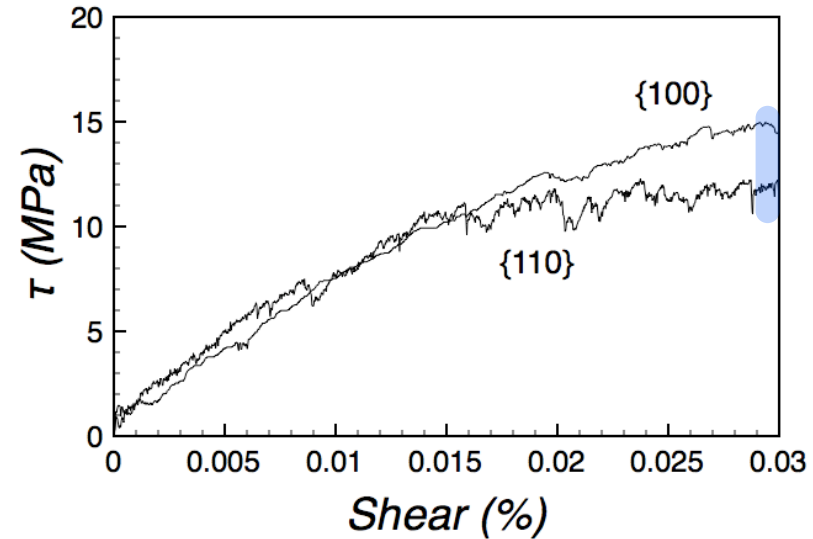
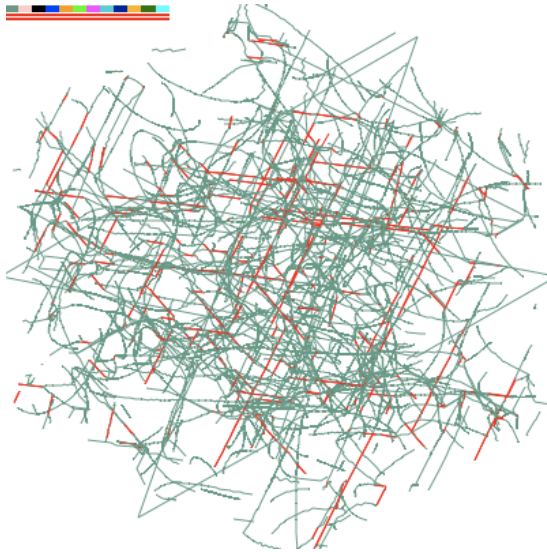
Following Pizzagalli et al. (2008)



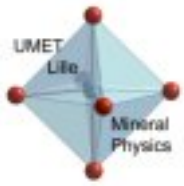
$$2U_k = (E_3 + E_4) - (E_1 + E_2)$$



DD results in Forest regime

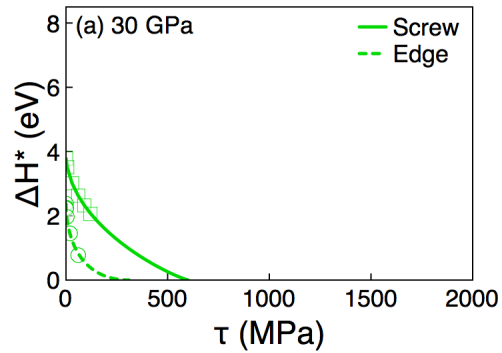


- $\sim 10^{12} \text{ m}^{-2}$ Dislocation density, equally allocated to the 6 slip systems, for both glide family
Dislocation density has to be consistent with the experimental value of τ_{μ} ($\sim 15 \text{ MPa}$)
- Strain rate chosen in the range of the dislocation dynamic behavior : strain has to be governed by forest interactions (dynamic behavior) not by dislocation velocity (quasi static behavior)

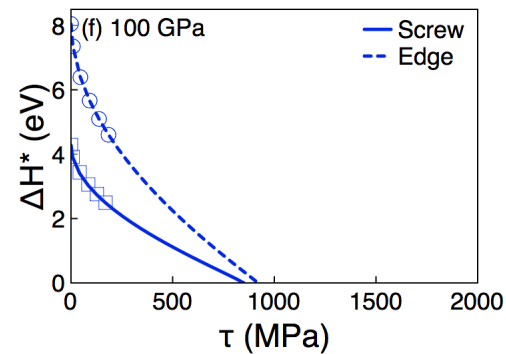
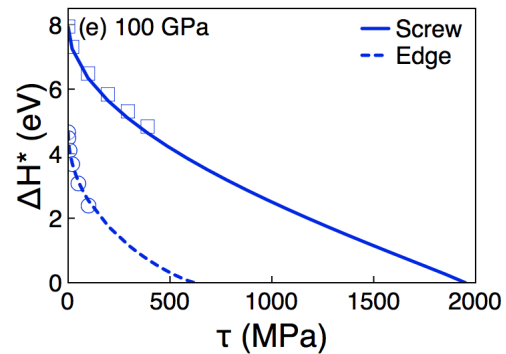
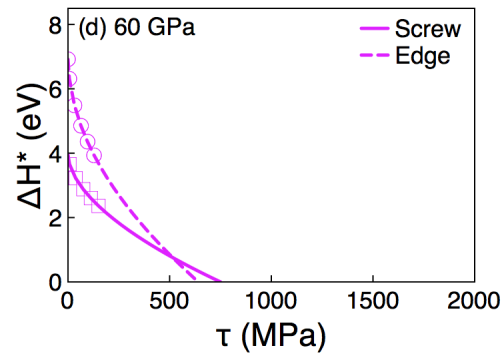
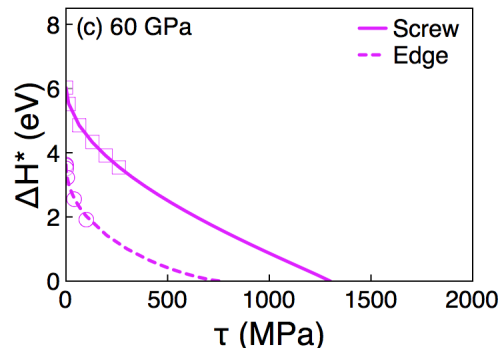
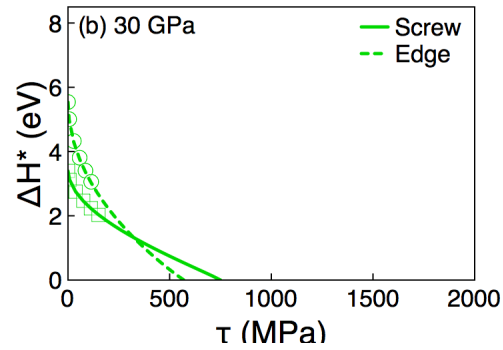


Kink pair nucleation enthalpy

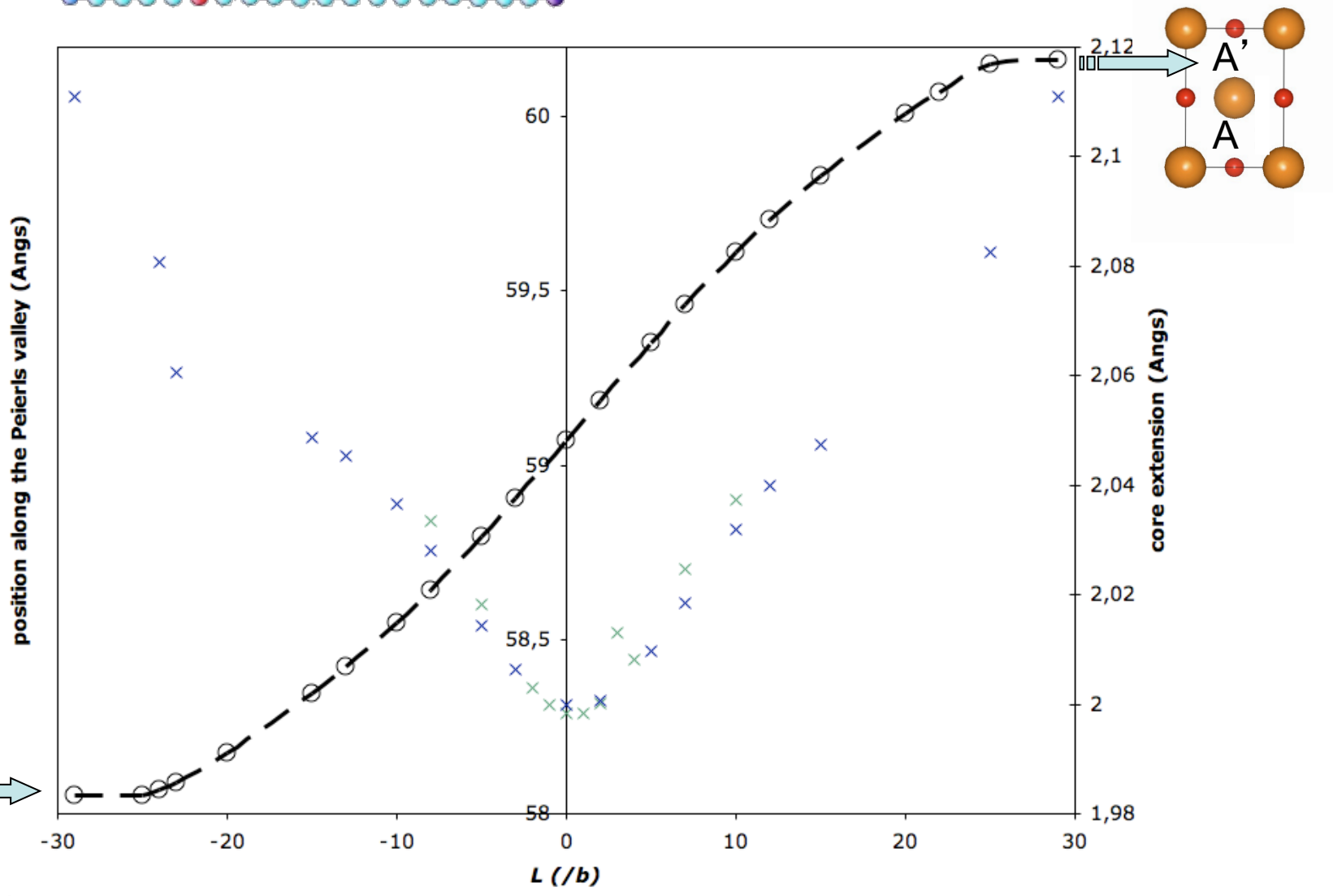
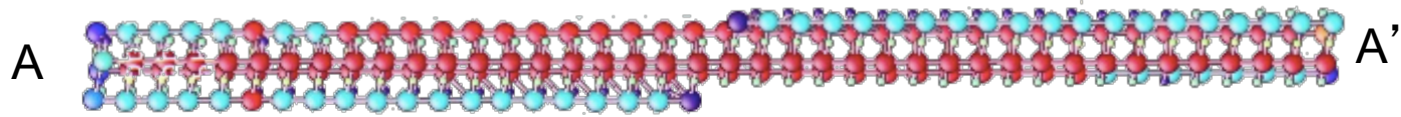
$1/2 \langle 110 \rangle \{110\}$



$1/2 \langle 110 \rangle \{100\}$



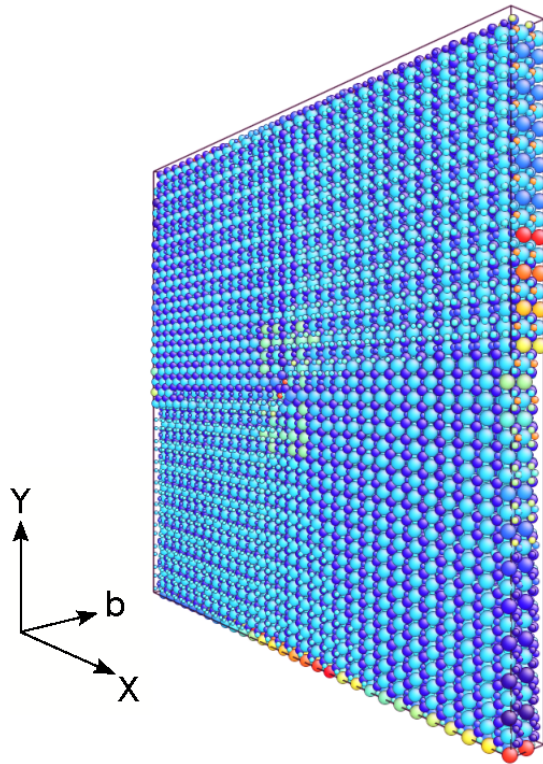
(Amodeo et al. 2012)



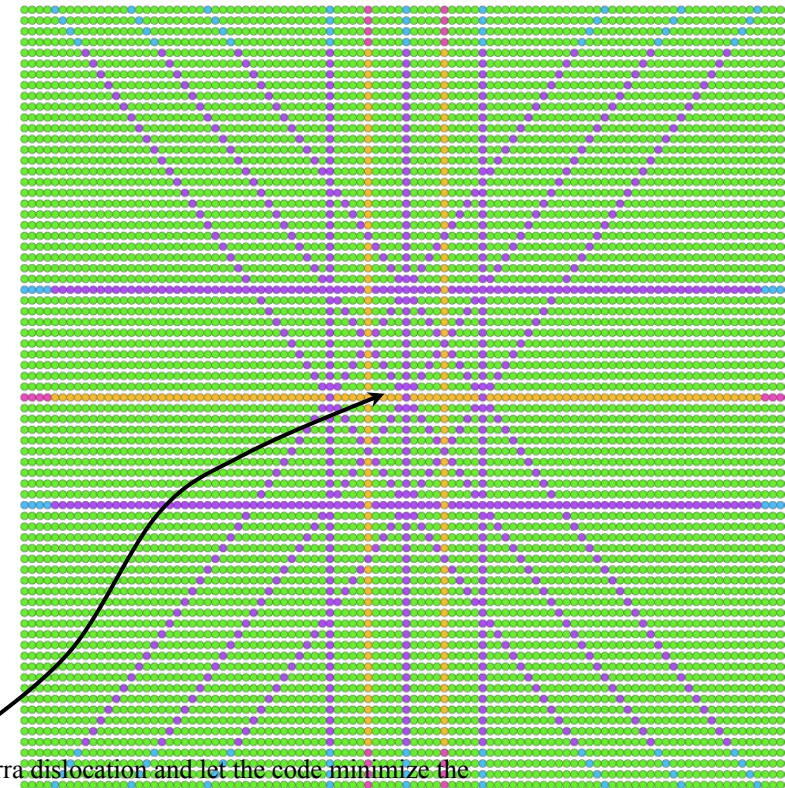
Dislocation core structures using either “cluster” approach or Peierls-Nabarro method

Atomistic “Cluster” approach

Single screw dislocation in a $300 \text{ \AA} \times 300 \text{ \AA} \times b$ cell with fixed region to elastic displacement field



Peierls-Nabarro-Galerkin method (Denoual 2004,2007)



Introduce a Volterra dislocation and let the code minimize the structure

Numerical examination of PN equations is performed using a nodal mesh, taking into account for a dislocation density in given plane P_i , interaction between dislocation density distributed in P_j , balanced with inelastic stacking fault (from gamma-surface) attributed to P_i plane

[100] screw dislocations: (010), (001), (011) γ -surfaces

[010] screw dislocations: (100), (001), (101) γ -surfaces